

# X-RAY ABSORPTION SPECTROSCOPY (XAS)



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July 29, 2024  
26<sup>th</sup> National School on Neutron  
and X-ray Scattering

## OUTLINE

- Background X-ray absorption spectroscopy (XAS)
- X-ray absorption near edge spectra (XANES) process
- XANES examples
- Extended X-ray absorption fine structure (EXAFS) fundamentals
- EXAFS examples

## 1920 PUBLISHED ABSORPTION EDGES

The K-characteristic absorption frequencies for  
The chemical elements magnesium to chromium.  
By Hugo Fricke

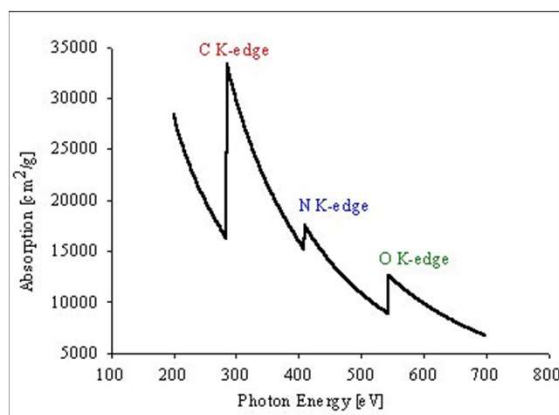
### Synopsis

**Absorption of X-rays.** – This paper contains an account of an experimental investigation concerning the **discontinuity in the x-ray absorption** corresponding to the K-series for the chemical elements from magnesium to chromium inclusively. The method followed was the same as that devised and employed by de Broglie. A specially designed vacuum spectrograph was used.

**Fine Structure of Absorption.** – The spectrograms show that the **discontinuity has a rather complex structure**, a result in advance of those obtained by earlier investigators. A photometric study of the plates was made in order to obtain a more accurate knowledge of the detailed structure of the absorption limits.

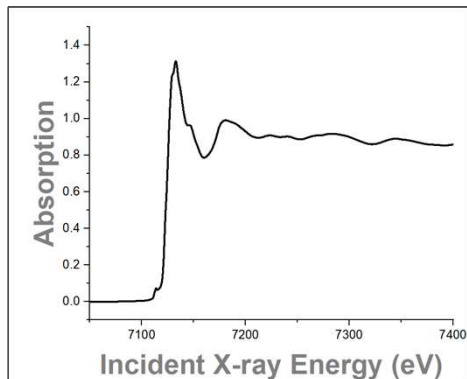
**Results.** These are recorded in tables which give for each element the wave-lengths of the different remarkable points in the structure of the discontinuities. The theoretical bearing of the new observations is briefly discussed.

## THE DISCONTINUITY IN THE X-RAY ABSORPTION CORRESPONDING TO THE K-SERIES



John (<https://physics.stackexchange.com/users/101660/john>). Why do we have the absorption edge?, URL (version: 2016-02-19): <https://physics.stackexchange.com/q/238105>

## THE DISCONTINUITY HAS A RATHER COMPLEX STRUCTURE



## 1971 UNDERSTANDING OF EXAFS

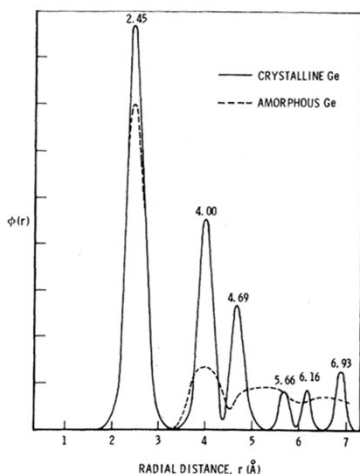


FIG. 2. Fourier transformation of the data of Fig. 1.  $\phi(r)$ , a radial structure function, compares amorphous and crystalline Ge. Numbers over the peaks indicate the measured distances in Å.

New Technique for Investigating Noncrystalline structures: Fourier Analysis of the Extended X-ray – Absorption Fine Structure

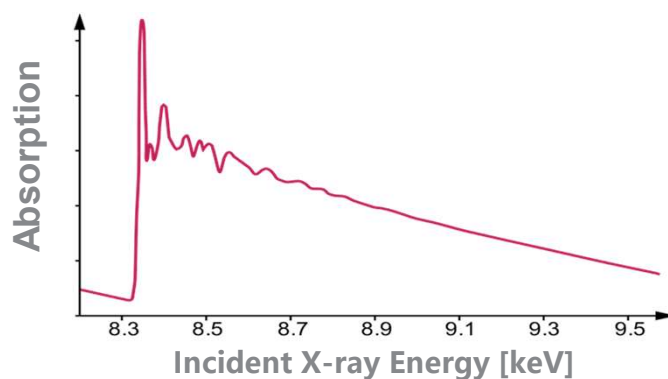
Dale E. Sayers, Edward A. Stern and Farrel W. Lytle  
Physical Review Letters 1971

<https://link.aps.org/doi/10.1103/PhysRevLett.27.1204>

## XANES AND EXAFS

XANES: x-ray absorption near edge structure

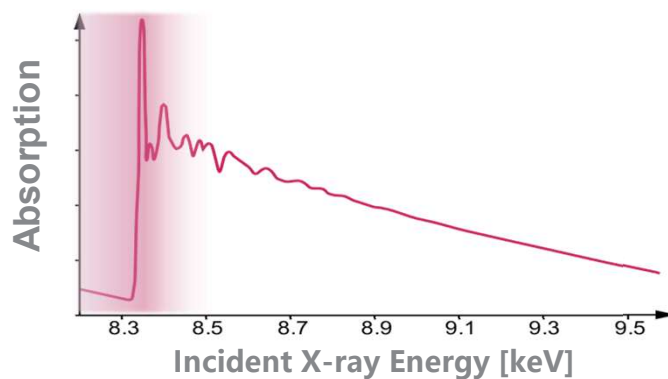
EXAFS: Extended x-ray absorption fine structure



## XANES AND EXAFS

XANES: x-ray absorption near edge structure

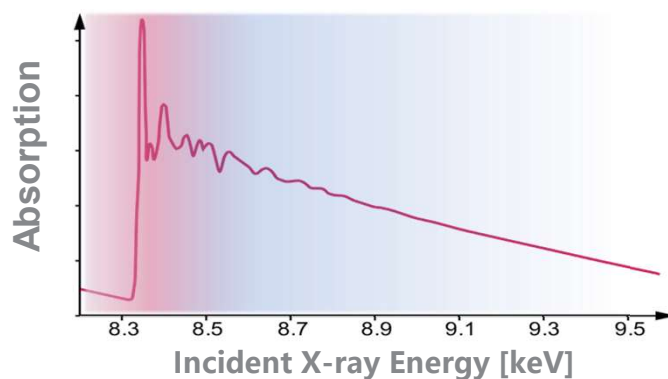
EXAFS: Extended x-ray absorption fine structure



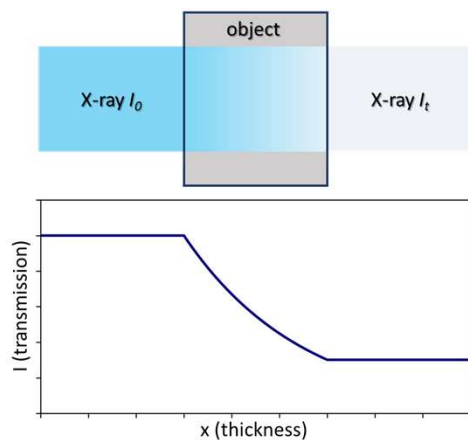
## XANES AND EXAFS

XANES: x-ray absorption near edge structure

EXAFS: Extended x-ray absorption fine structure



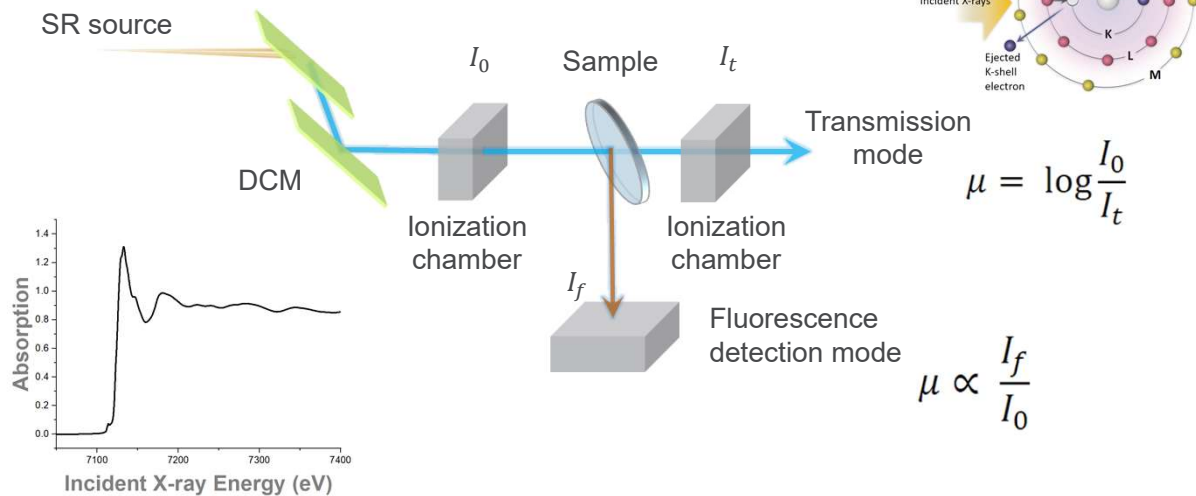
## BEER'S LAW: ABSORPTION OF X-RAYS BY MATTER



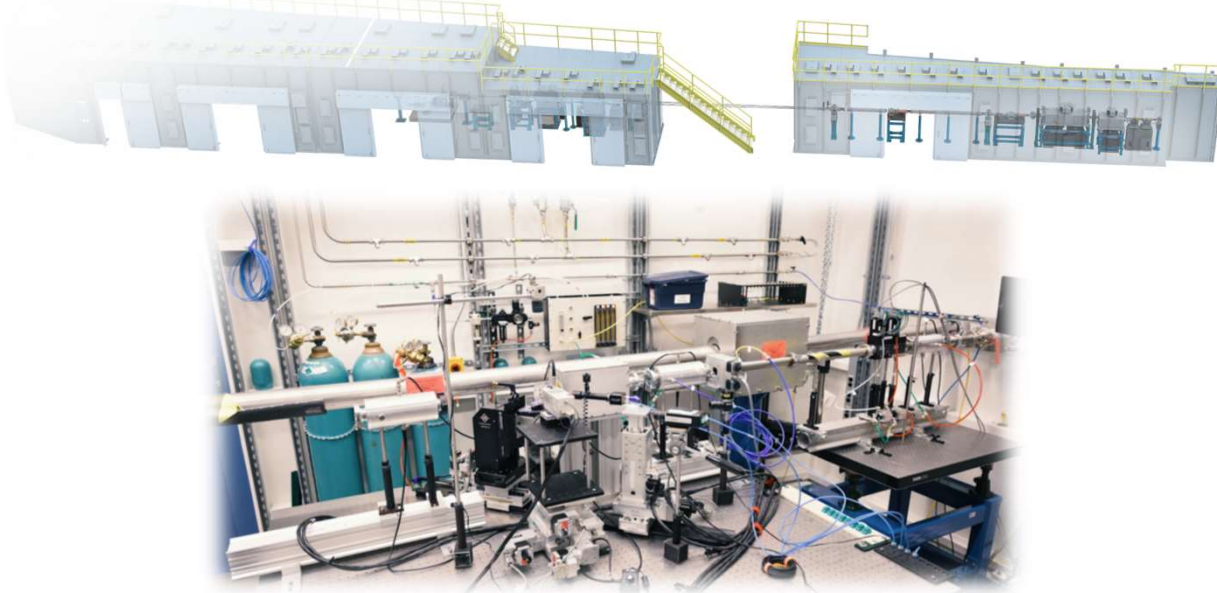
$$I_t = I_0 e^{-\mu x}$$

- $\mu x$ : absorption length of a material
- One absorption length,  $I_t = 37\% I_0$
- Two absorption lengths,  $I_t = 13\% I_0$

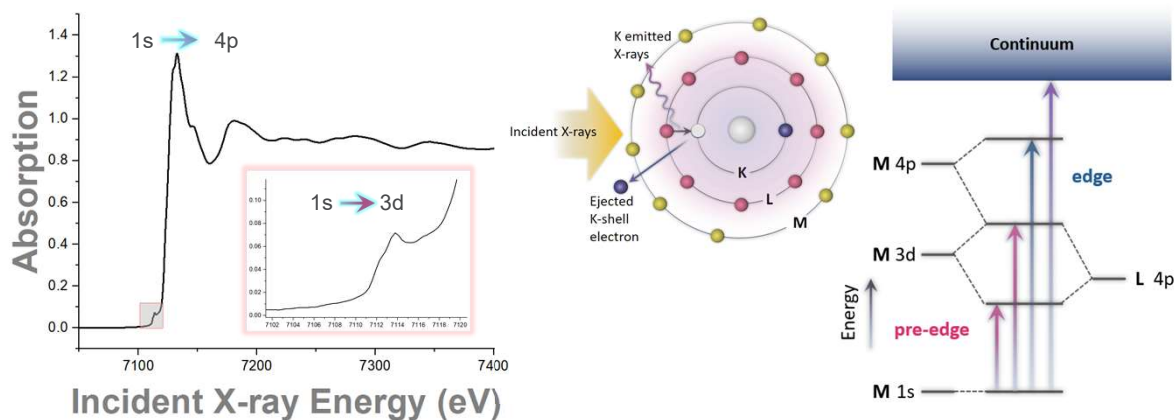
# MEASUREMENT OF X-RAY ABSORPTION COEFFICIENT



# EXPERIMENTAL SETUP AT 25-ID-C



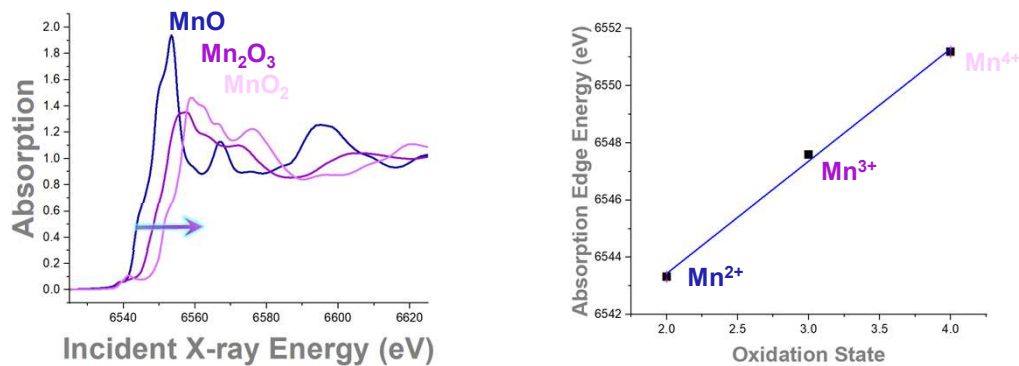
## METAL K-EDGE XANES



- Absorption edge: dipole  $1s \rightarrow 4p$  transition ( $\Delta l = \pm 1$ )
- Pre-edge: mixing of 3d-4p opens  $1s \rightarrow 3d$  transition

## OXIDATION STATE OF MN OXIDES

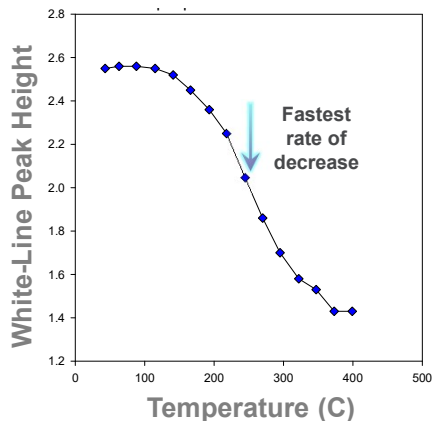
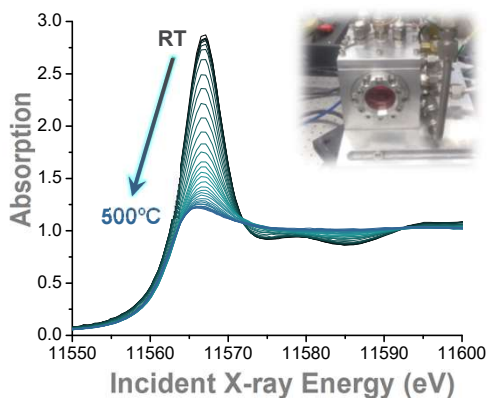
### Mn K-edge XANES



Many elements show significant absorption edge shifts with oxidation state.

## PT XANES DURING IN SITU REDUCTION

### Direct in situ measurement of Pt reduction



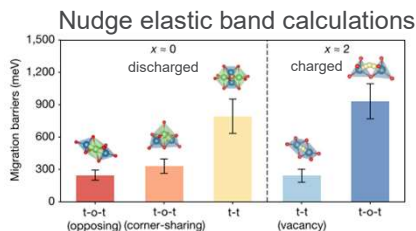
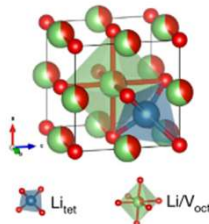
- Pt  $L_3$ -edge  $p \rightarrow d$  transition
- Pt electrons:  $[\text{Xe}]4f^{14}5d^6s^1$
- Pt white line intensity decreases as temperature increases due to 5d electrons filling
- Rate of decrease is fastest at  $\sim 270^\circ\text{C}$ .

## FAST-CHARGING LITHIUM-ION BATTERIES

### Haodong Liu, et al., Nature, 2020

Unique disordered rock salt (DRS) anode reversibly cycles two  $\text{Li}^+$  at a low 0.6 volts versus a  $\text{Li}/\text{Li}^+$  reference cathode reducing the short-circuit risk due to Li dendrite growth. 40% capacity can be delivered in 20 seconds!

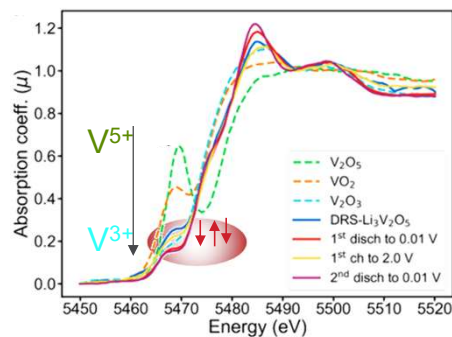
High performance is due to Li intercalation mechanism with low energy barriers and small volume change which cause V to change oxidation state.



T-O-T  $\text{Li}^+$  in t-site hops to neighboring occupied o-site, and Li in o-site hops to empty t-site.

T-O-T  $\text{Li}^+$  in t-site hops through and empty O-site to another t-site

### 9-BM V K-edge XANES



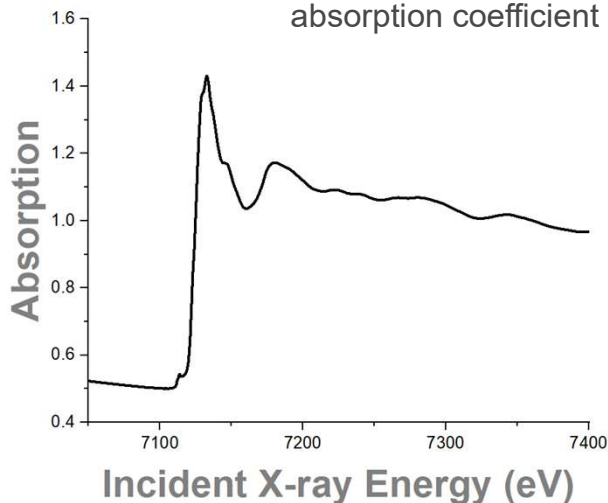
Low energy range at 9-BM enables V K-edge XANES spectra to confirm oxidation state changes with charge and discharge. V in pristine DRS is mixture of  $\text{V}^{4+}$  and  $\text{V}^{3+}$ , after discharge the V oxidation state is less than  $\text{V}^{3+}$ , oxidation state switches back and forth during the 1<sup>st</sup> charge and 2<sup>nd</sup> discharge showing highly reversible V oxidation state change.

The team includes 26 authors with expertise in electrode chemistry, materials synthesis, neutron diffraction, in-situ XRD ICP-OES, STEM, XAS, SEM, XPS, and DFT.



## DEFINITION OF EXAFS $\chi(E)$

Normalized oscillatory part of the absorption coefficient

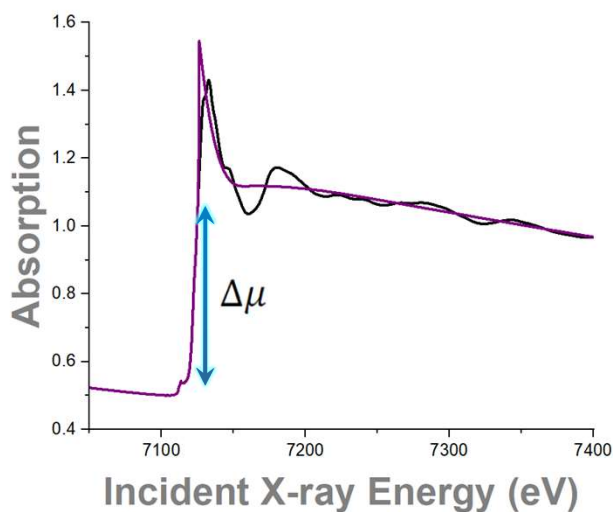


$$\chi(E) = \frac{\mu(E) - \mu_0(E)}{\Delta\mu(E)}$$

Measured absorption coefficient in transmission or fluorescence

$$\mu = \log \frac{I_0}{I_t} \quad \mu \propto \frac{I_f}{I_0}$$

## DEFINITION OF EXAFS: EDGE STEP

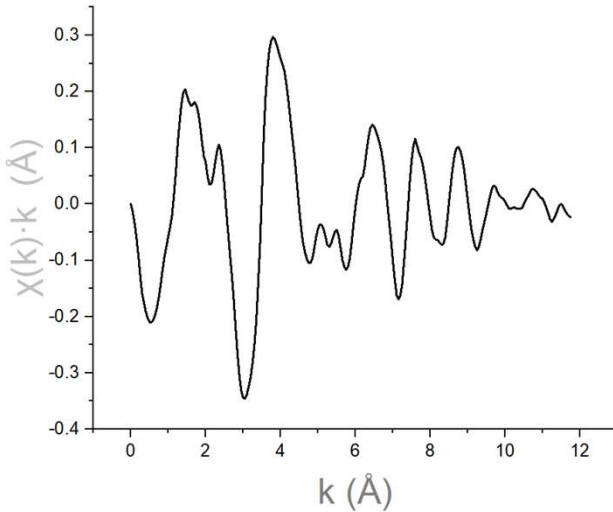


$$\chi(E) = \frac{\mu(E) - \mu_0(E)}{\Delta\mu(E)}$$

Smoothly varying background function, representing the absorption without fine structure

Normalized by the edge step height

## DEFINITION OF EXAFS: $\chi(k)$



$$\chi(E) = \frac{\mu(E) - \mu_0(E)}{\Delta\mu(E)}$$

$$k^2 = 2 m_e(E-E_0)/ \hbar$$

## FERMI'S GOLDEN RULE

$$\mu(E) \propto |\langle i|H|f\rangle|^2$$

$\langle i|$  Initial State: atom with core electron

$H$  Interaction term: incident x-ray

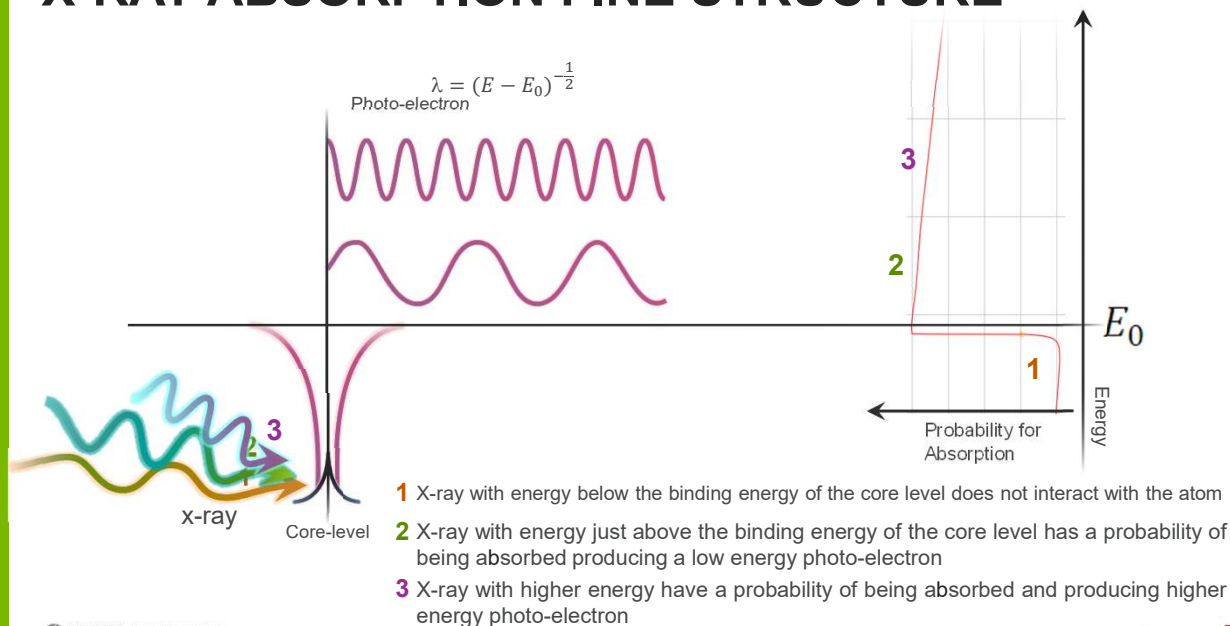
$|f\rangle$  Final State: atom with core hole, photo-electron

- Transition between two quantum states
- Initial state is well localized at the absorbing atom
- Final state is not, but can be written in terms of two parts

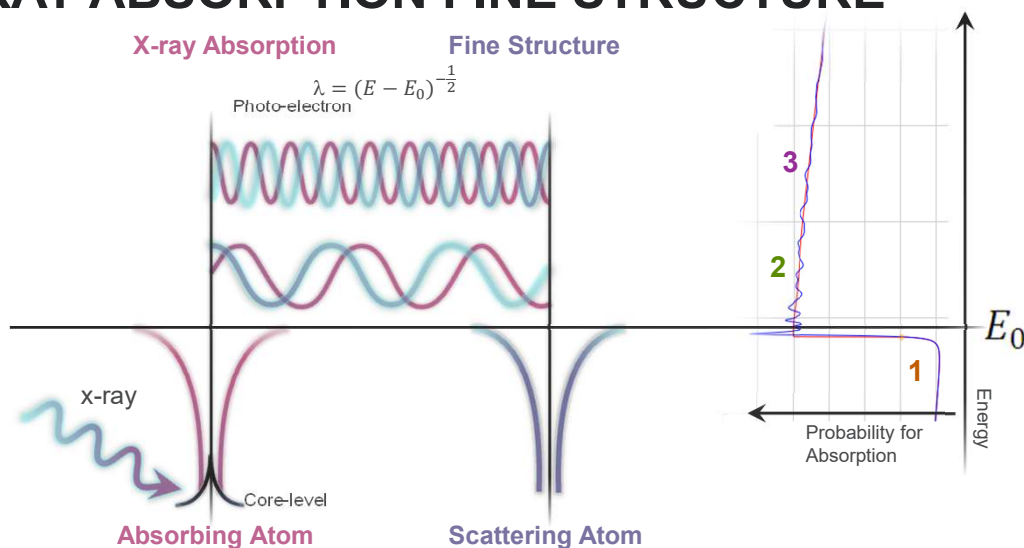
$$|f\rangle = |f_0\rangle + |\Delta f\rangle$$

adsorbing      neighboring

## X-RAY ABSORPTION FINE STRUCTURE

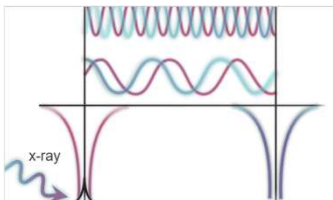


## X-RAY ABSORPTION FINE STRUCTURE



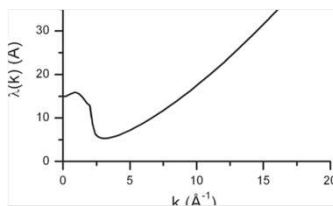
Interference between outgoing and scattered photoelectron at the absorbing atom causes modulations in the probability for absorption.

## ADDITIONAL EXAFS DEPENDENCIES

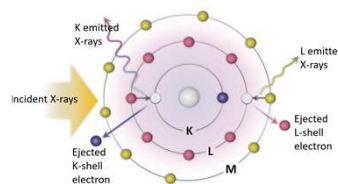


- Atomic Phase shift has two parts. One part from the absorbing atom and another part from the scattering atom.

$$\chi(k) \propto \frac{F(k)}{k^2 R^2} \sin(2kR + \delta(k))$$



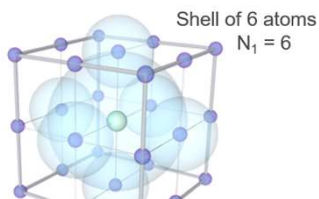
- Mean free path of the photoelectron depends on k causes EXAFS to decrease at high k and contributing to complexity in XANES region at low k  $e^{-\frac{2R}{\lambda_j}}$



- Passive electron reduction factor: The initial and final states include all the passive electrons of the absorbing atom.

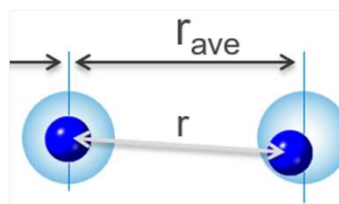
$$S_0^2 = |\langle \phi_f^{N-1} | \phi_i^{N-1} \rangle|^2$$

## AVERAGE OVER SHELLS OF ATOMS



- EXAFS is a sum of all the scattering events of the photoelectron
- Convenient to group "shells of atoms"

$$\chi(k) = \sum_i N_i \chi_i(k)$$



- Not all the atoms in the shell are at the same average distance from the absorbing atom.
- Mean-square displacement of the half path length  $e^{-2k\sigma^2}$

## THE EXAFS EQUATION

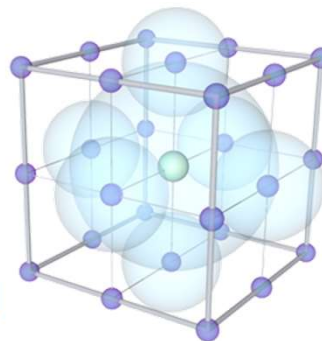
$$\chi(k) = \sum_i \chi_i(k)$$

with

$$\chi(k) = \sum_j \frac{N_j S_0^2 F_j(k) e^{-2k^2 \sigma_j^2} e^{-\frac{2R}{\lambda}}}{k R_j^2} \sin[2kR_j + \delta_j(k)]$$

$$R_i = R_0 + \Delta R$$

$$k^2 = 2 m_e (E - E_0) / \hbar$$



### Theoretically calculated values

$F_i(k)$  effective scattering amplitude

$\delta_i(k)$  effective scattering phase shift

$\lambda(k)$  mean free path

### Starting values

$R_0$  initial path length

### Parameters determined from a fit to data

$N_j$  degeneracy of path

$S_0^2$  passive electron reduction factor

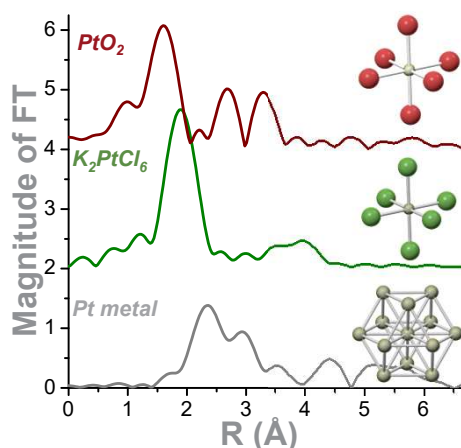
$\sigma_j^2$  mean squared displacement of half-path length

$E_0$  energy shift

$\Delta R$  change in half-path length

## PT EXAFS OF REFERENCE MATERIALS

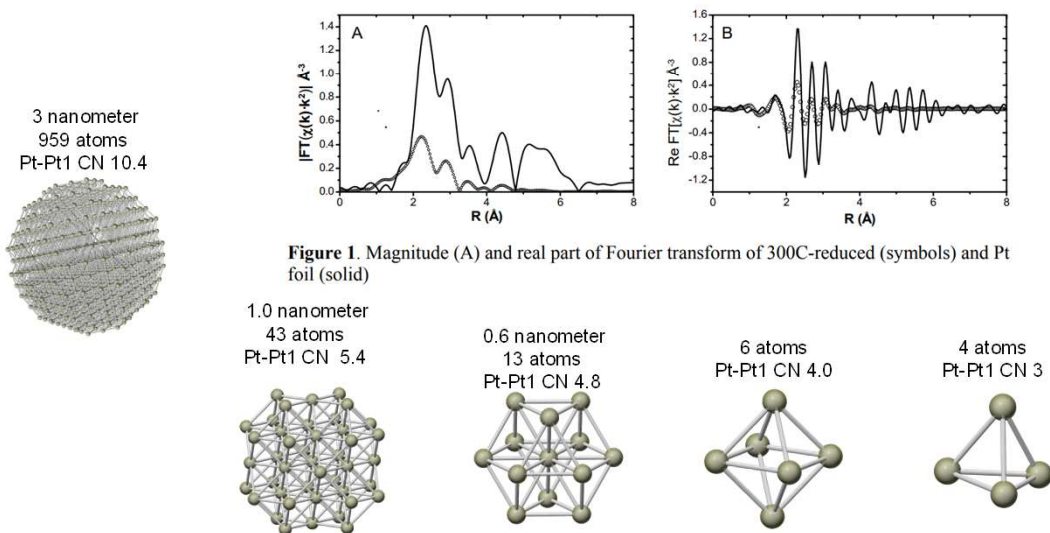
**Pt EXAFS: unique information about the average local atomic environment of Pt**



PtO <sub>2</sub>		
Neighbor	Number	Distance (Å)
Pt-O	6	2.07
Pt-Pt	6	3.10
K <sub>2</sub> PtCl <sub>6</sub>		
Neighbor	Number	Distance (Å)
Pt-Cl	6	2.32
Pt-K	4	4.22
Pt metal		
Neighbor	Number	Distance (Å)
Pt-Pt	12	2.77
Pt-Pt	6	3.92

Pt-O, Pt-Cl, and Pt-Pt signals are unique and are readily distinguished.

# PLATINUM EXAFS: COMPARISONS WITH FOIL



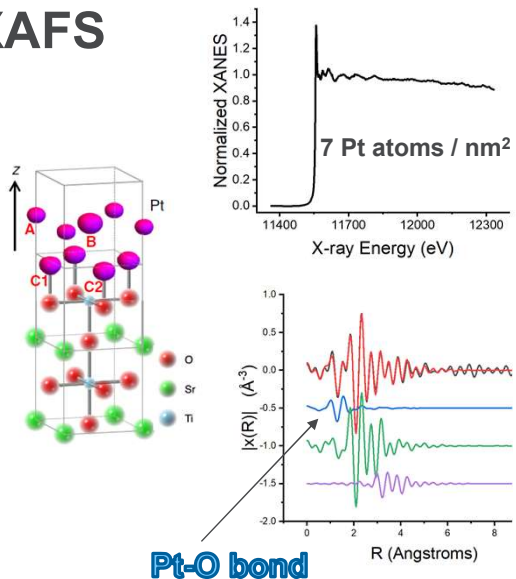
<https://iopscience.iop.org/article/10.1088/1742-6596/430/1/012061/pdf>

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## GRAZING INCIDENCE PT EXAFS

### Pt monolayer on SrTiO<sub>3</sub> (001) substrate

- Oxide supported noble metal nanoparticles are widely used as heterogeneous catalysts; playing an important role for the societal shift from a fossil-fuel to renewable energy sources.
- Studies of highly diluted monolayer and single atom catalyst are difficult, but necessary with catalyst development at the atomic-scale.
- Initial measurements show full EXAFS scans can be used to determine interfacial Pt-O bond



## EXAFS STUDY OF $Nb_3Sn$ SUPERCONDUCTORS

Heald S. et al., *Scientific Reports* 2018; Tarantini C, et al. *Superconductor Science and Technology* 2019

### Challenge

- $Nb_3Sn$  proposed for future accelerator upgrades, but needs improved properties
- Doping can offer improvement, but optimization needs better understanding

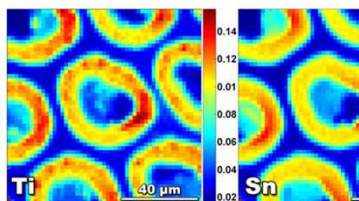
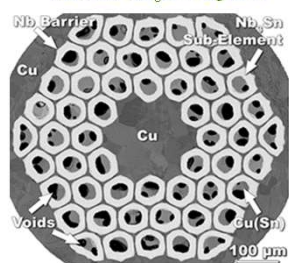
### EXAFS

- Determine dopant lattice location.
- When combined with other results offered key insights into the role of dopants

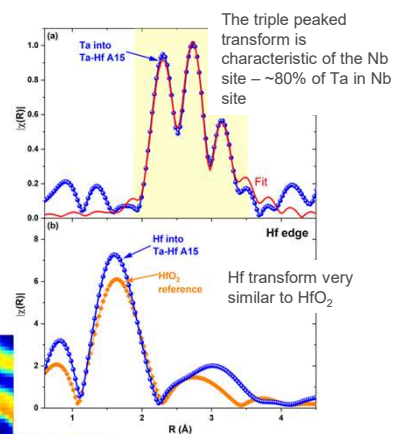
### Result

- Ti, Ta, and Hf dopants studied
- Determined Ta dopant increased antisite disorder with beneficial results
- Hf formed  $HfO_2$  nanoparticle pinning sites
- Combined Ta and Hf doping offers promising route to meeting the needs of future accelerators.

Microprobe needed to measure narrow  $Nb_3Sn$  regions



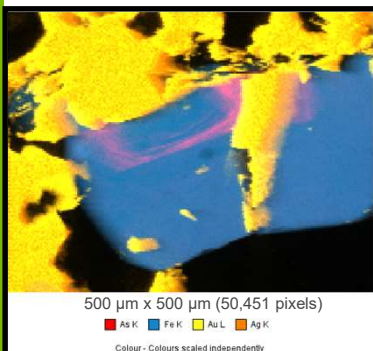
Fourier Transformed EXAFS



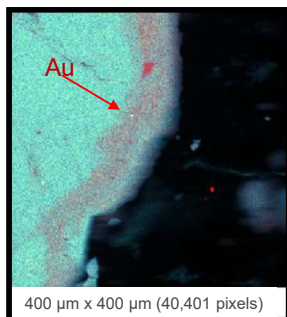
## BONANZA GOLD MECHANISM

### Microprobe XRF and Spectroscopy

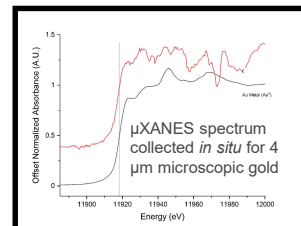
Extremely high-resolution SR- $\mu$ XRF mapping of arsenian pyrite reveals that bonanza-style gold mineralization was caused by gold flocculation from electron transfer near arsenic-rich bands.



The distribution of electrum (Au,  $\blacksquare$  + Ag,  $\blacksquare$ ) on the edges of corroded pyrite grains (Fe,  $\blacksquare$ ) with As banding (As,  $\blacksquare$ ) as fine as  $<2 \mu m$  (single pixel thickness!)



Microscopic metallic gold grain (2 pixels wide,  $\blacksquare$ ) within the As band (As,  $\blacksquare$ ) on the edge of a pyrite grain (Fe,  $\blacksquare$ ).

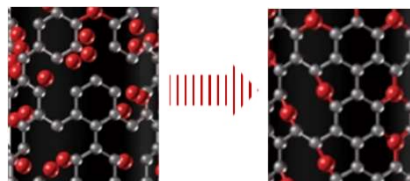


Microscopic gold within arsenian pyrite growth zone is metallic  $Au^0$  and not lattice bound  $Au^{+1}$

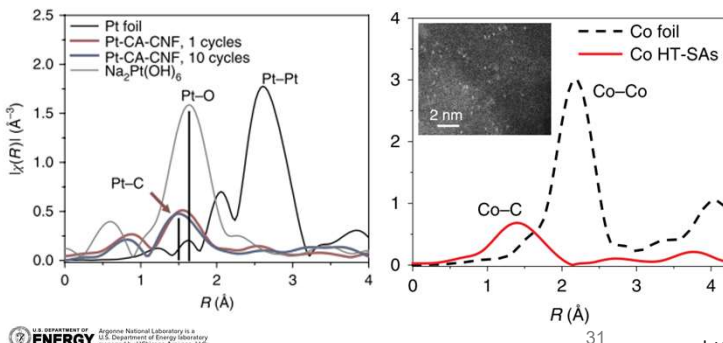
# HIGH TEMPERATURE SHOCKWAVE STABILIZED SINGLE-ATOM CATALYSTS

Yao, et al., *Nature Nanotechnology*, 2019

Novel general manufacturing route for single-atom catalysts. Shockwaves with controlled 1,500K high-temp and 55ms short on-state followed by longer off-state. Process ensures dispersion and anchoring of metal atoms on substrate defect sites that are highly stable. Overcoming conventional catalyst deactivation mechanism through metal atom agglomeration.



9-BM Pt and Co EXAFS



Pt and Co EXAFS at 9-BM shows the typical Metal-C signal that is unchanged from the 1-cycle to the 10-cycle. There is no evidence of agglomeration which would present itself as a strong signal as shown at 2.5 Å for metallic foil.

The team includes 20 authors with expertise in shockwave materials synthesis, S/TEM, In situ ETEM, Raman spectroscopy, surface area measurements, ICP-MS, EXAFS, MD and DFT simulations, as well as catalysis performance measurements. Tianpin Wu from 9-BM is corresponding author.

# STRUCTURE OF LUMINESCENT PROTEIN-STABILIZED GOLD CLUSTERS

EXAFS with DFT interpretation

Chemical Science

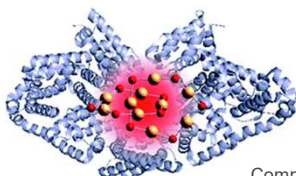
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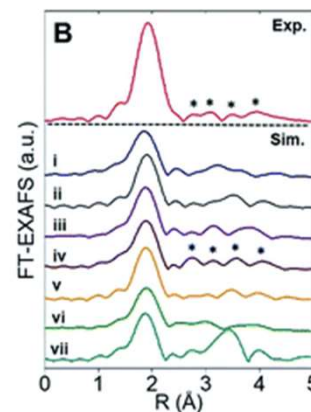
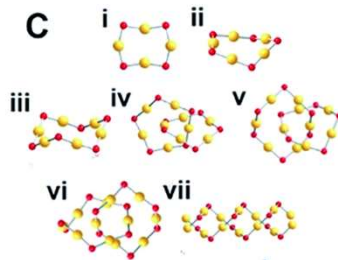
Structure and formation of highly luminescent protein-stabilized gold clusters†

D. M. Chevrier,<sup>1</sup> V. D. Thanthirige,<sup>2</sup> Z. Luo,<sup>2</sup> S. Driscoll,<sup>2</sup> P. Cho,<sup>2</sup> M. A. MacDonald,<sup>2</sup> Q. Yao,<sup>2</sup> R. Guda,<sup>2</sup> J. Xie,<sup>2</sup> E. R. Johnson,<sup>2</sup> A. Chatt,<sup>2</sup> N. Zheng,<sup>2</sup> and P. Zhang<sup>2\*</sup>

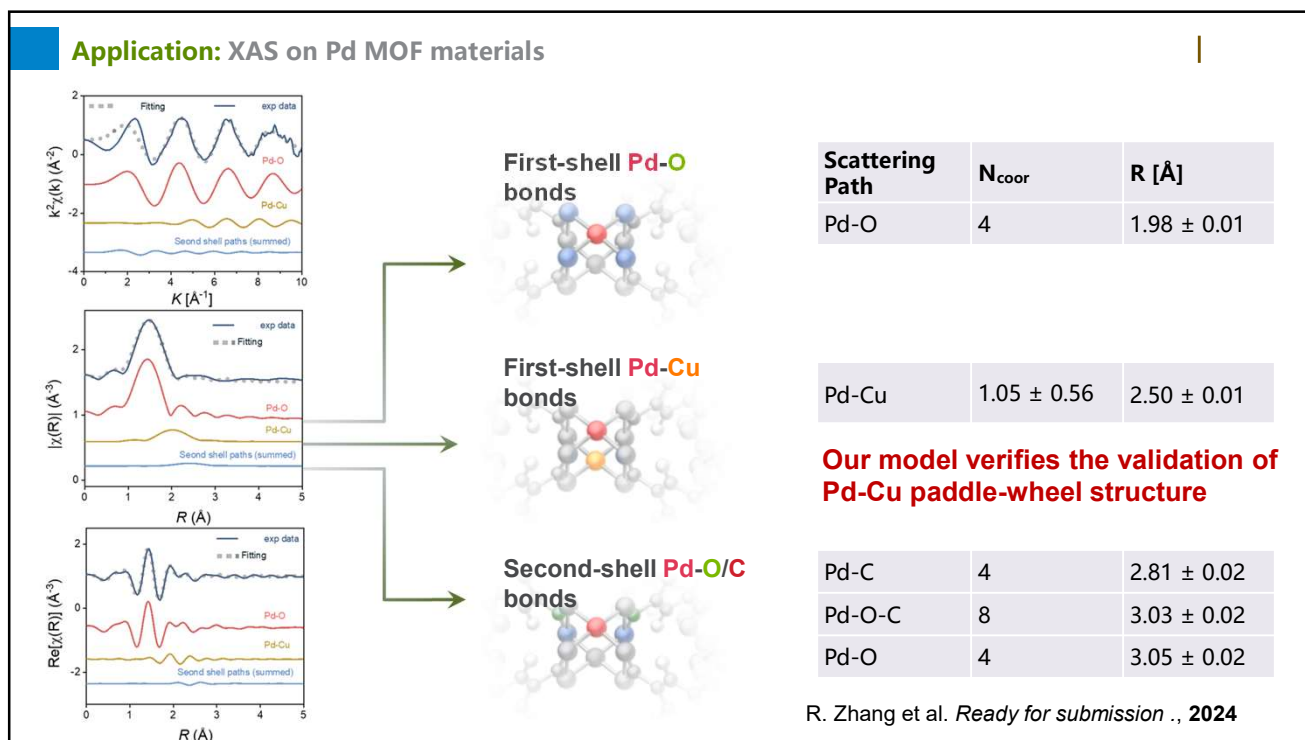
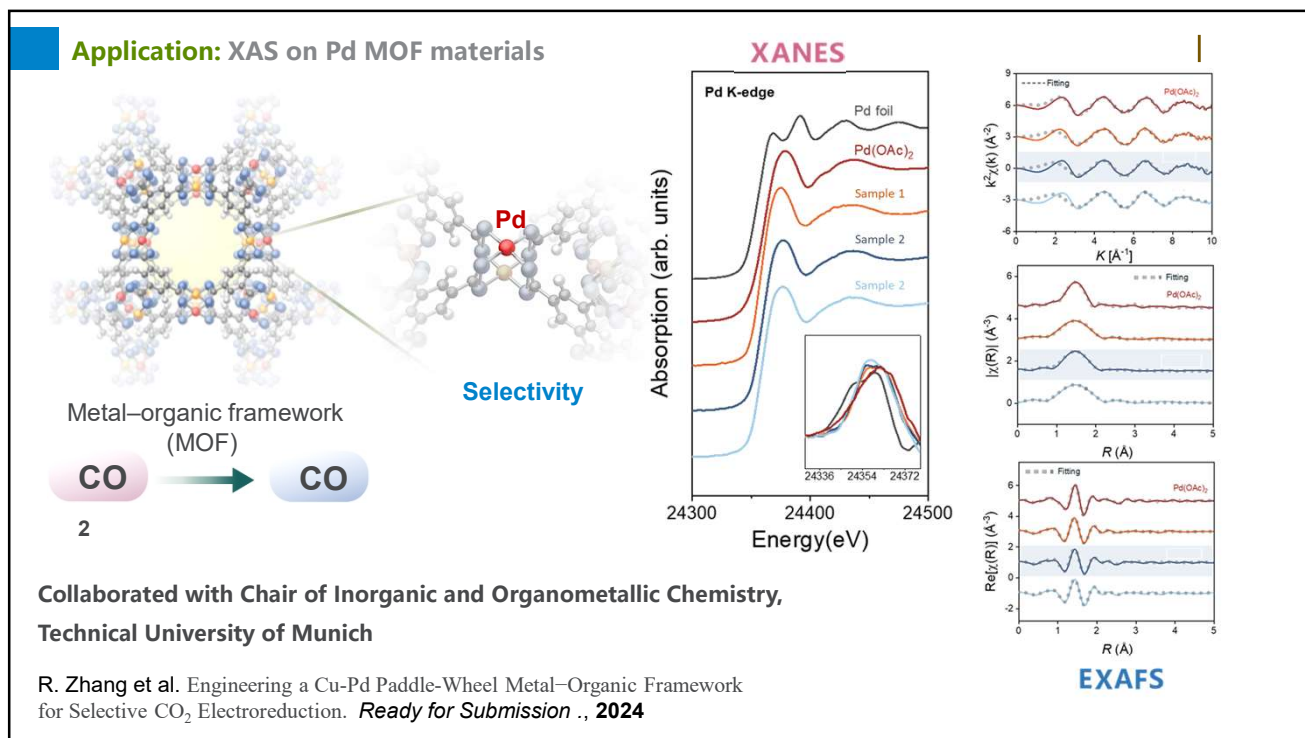


Comparison between DFT simulated structure and measured spectra give insights into the gold cluster morphology

<https://doi.org/10.1039/C7SC05086K>







# Q & A

Kelly, S. D.; Hesterberg, D.; Ravel, B.; Analysis of Soils and Minerals Using X-Ray Absorption Spectroscopy. Methods soil Anal. Part 5. Mineral. methods 2008, 5, 387–464.