



ORNL is managed by UT-Battelle, LLC for the US Department of Energy

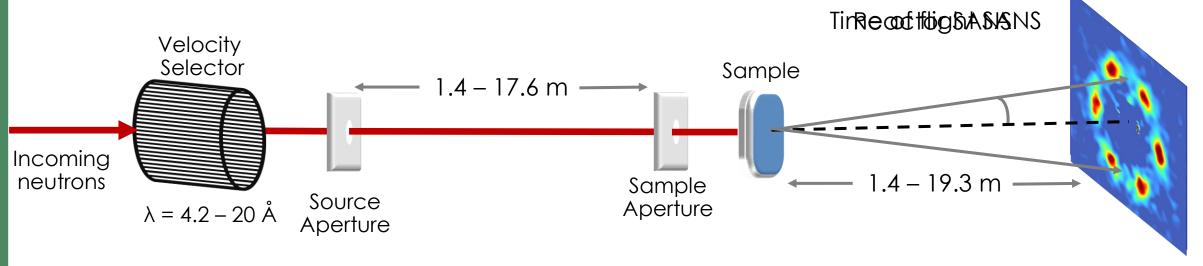


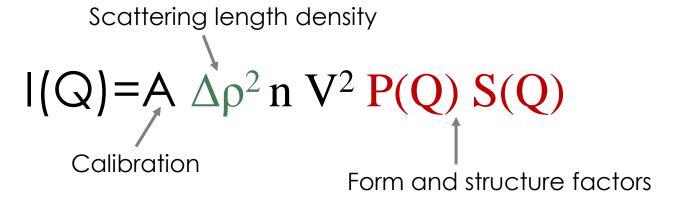
Outline:

- What is SANS?
- Instruments at ORNL
- Sample Environment capabilities
- Science Cases
 - Polymers
 - Magnetic ordering (diffraction)
 - Engineering materials
- New Directions



Small-angle neutron scattering (SANS)





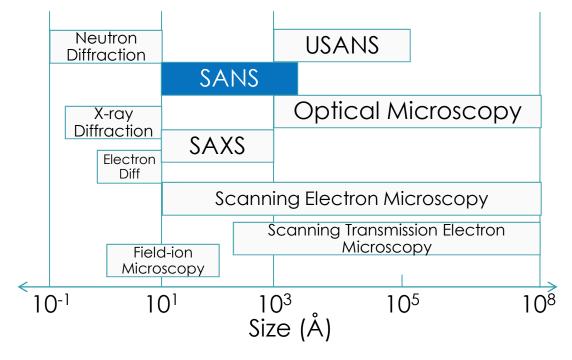
Probes lengths of 0.5 - 700 nmUsing angles of $\theta = <0.1$ to 45°

A powerful tool for studying nanophase bulk materials

SANS with other techniques

Microscopy: Direct but limited



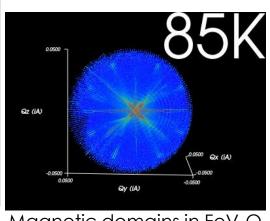




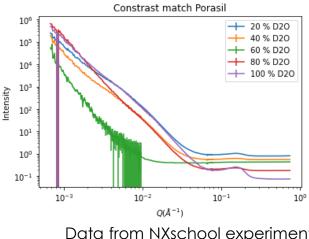
SAS: Indirect and model-dependent but in-situ

Why Neutrons?

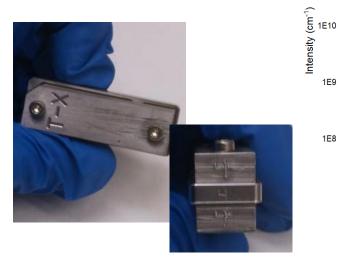
- Low Energy (E=80meV for 1 Å neutrons vs. 12.42 keV for x-rays. No radiation damage.)
- High Penetration of cold neutrons bulk samples
- Large difference in the scattering cross-section for the hydrogen and deuterium- contrast variation capability.
 - Solute or solvent can be deuterated to vary the contrast ($\rho_{H2O} = -0.56 \times 10^{10} \text{cm}^{-2}$, $\rho_{D2O} = 6.334 \times 10^{10} \text{cm}^{-2}$)
 - Study of multicomponent systems through selective deuteration and contrast matching with H/D mixtures.
- Sensitive to substantial difference in scattering length density in transition metals
- MAGNETISM.
 - Neutron has a magnetic moment and spin which give contrast



Magnetic domains in FeV₂O₄ L. Kish et al in preparation



Data from NXschool experiment



q (Å⁻¹) Copper precipitate growth in steel

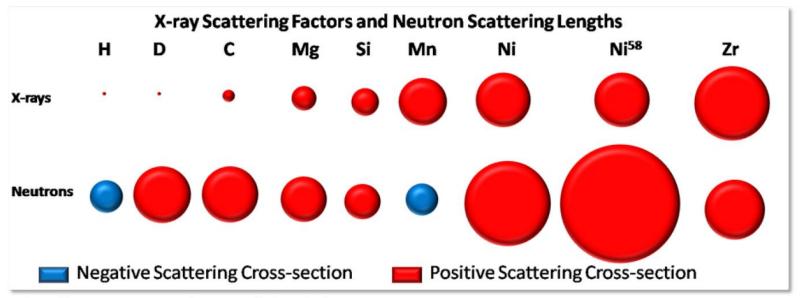
0.1

Neutron and x-ray scattering cross-sections

X-ray and neutron scattering are essentially the same concept, except...

X-rays scatter from electrons

Neutrons scatter from nuclei

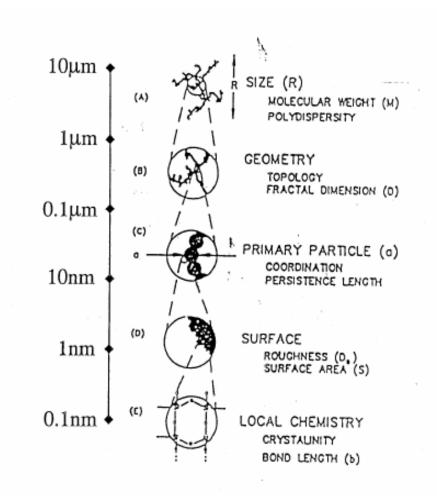


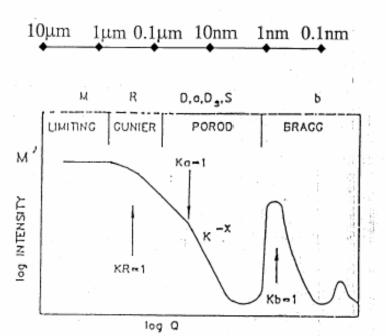
http://www.ncnr.nist.gov/resources/n-lengths/ http://www.isis.rl.ac.uk/ISISPublic/reference/Xray_scatfac.htm

C. Metting, Dissertation University of Maryland 2011



SANS from heirarchical structures





SANS is a structural technique that probes a wide array of length scales

SANS Suite



ORNL SANS instrument specifications.

	GP-SANS	Bio-SANS	EQ-SANS	USANS
Moderator	Supercritical hydrogen	Supercritical hydrogen	Coupled, supercritical hydrogen	Decoupled, poisoned hydrogen
Source	Reactor, continuous	Reactor, continuous	Spallation, 60 or 30 Hz, TOF	Spallation, 60 Hz, TOF
Sample-to-detector distance	1.1–20 m	1.1–15.5 m	1.0–9 m	N/A
Beam size	Up to 20 mm diameter	Up to 20 mm diameter	Up to 15 mm diameter	Up to 40 mm by 40 mm
Q range	7×10^{-4} to 1Å^{-1}	$9 \times 10^{-4} \text{ to } 1 \text{ Å}^{-1}$	$2 \times 10^{-3} \text{ to } 6 \text{ Å}^{-1}$	$7 \times 10^{-6} \text{ to } 5 \times 10^{-3} \text{ Å}^{-1}$
Incident wavelength	4–25 Å	6–25 Å	0.5–20 Å	3.6, 1.8, 1.2 and 0.9 Å
$\Delta \lambda / \lambda$	9–45%	9–45%	<1-20%†	Monochromated with crystals
Detector type	³ He LPSD array	³ He LPSD array	³ He LPSD array	³ He

[†] The resolution of EQ-SANS is wavelength dependent and also depends on the binning used in the software. The upper end of the range indicated is for the shortest wavelength used in the current data reduction.

Sample Environments:

Current Sample environment options:

- A suite of changers that can be configured to hold from 8 -18 samples
- Peltier system that goes from -10 to 120 °C with 0.01 degree control and holds 12 samples.
- HiDRA load frame; rheometer
- Shared furnace capable of temperature up to 200 °C
- New controlled atmosphere furnace and vacuum furnace to > 1000 °C
- Horizontal 11 T recondensing magnet allowing users to utilize 11 T at 30 mK. (3/2016), 5 T horizontal open bore magnet (5/2017) and 8 T vertical magnet with temps 30 mK to 300 K
 - Electric field and current can also be applied to the samples
- SANS dedicated cryostat with sapphire windows
- Pressure cells (McHugh 2kBar and 1kBar changer)

Future upgrades:

- Strain cell to apply in-situ compression or tension (11 T and 5 T magnets)
- In-situ reaction vessel
- Load frame for stretching polymers







Science cases

- Diblock copolymers
- Skyrmion in thin films
- Novel Steels for reactor





Influence of Cleavage of Photosensitive Group on Micellization and Gelation of a Doubly Responsive Diblock Copolymer

Lilin He¹, Bin Hu², Daniel M. Henn², and Bin Zhao²

¹ Large Scale Structure Group, Neutron Scattering Division, Oak Ridge National Laboratory

² Department of Chemistry, University of Tennessee: Knoxville

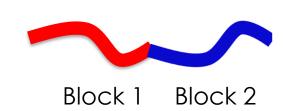


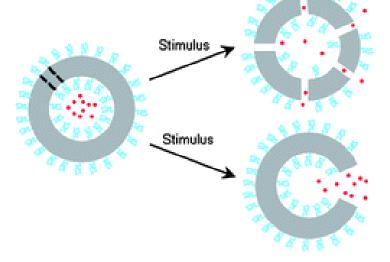




Influence of Cleavage of Photosensitive Group on Micellization and Gelation of a Doubly Responsive Diblock Copolymer

Block Copolymers have wide technical applications owning to their capabilities to self-assemble into nanostructures under different conditions.

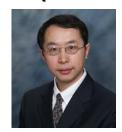




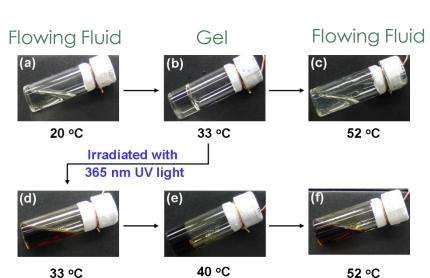
Controlled release example

Challenge: To precisely **tune** and **control** the molecular characteristics of the block copolymers under appropriate conditions.

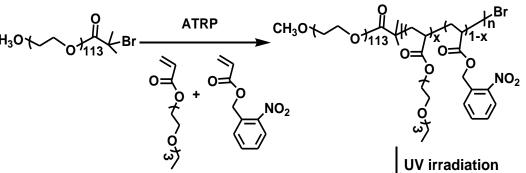
Thermal- and light-sensitive diblock copolymer PEO-b-P(TEGEA-co-NBA)



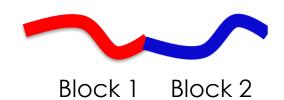
Prof. Bin Zhao, UTK



- Brown-red color due to the photochemical reaction
- Multiple transitions

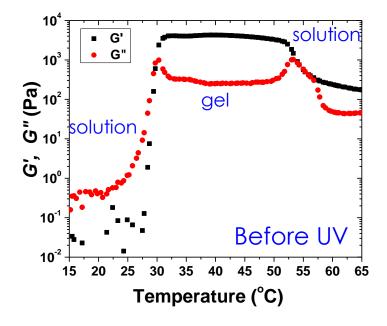


- "living"/controlled radical polymerization
- Irradiated for 6 days with 365nm UV light

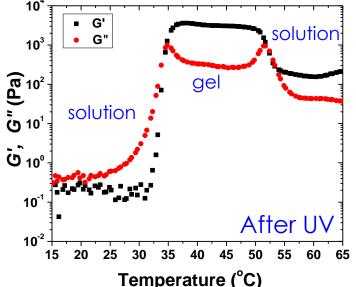




Rheological Results



- Dynamic storage modulus G', loss modulus G'', versus temperature
- 25wt % D_2O solution of PEO-b-P(TEGEA-co-NBA).
- A heating rate of 3°C/min.
- A strain amplitude of 1 %
- An oscillation frequency of 1 Hz.



- Reversible sol-gel-soft gel transitions is achieved;
- Gel is composed of packed micelles;
- UV irradiation leads to a narrower gel state window.

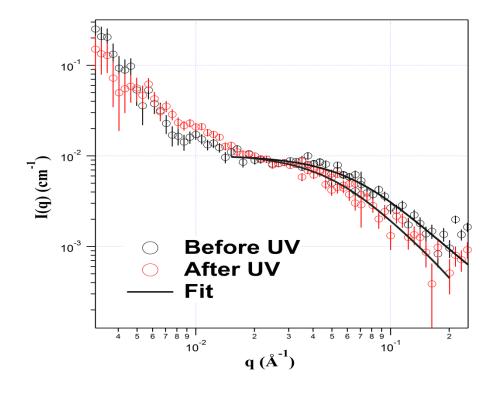
Rheological Properties



Microstructure



SANS results: Unimer State



Guinier Approximation at low-q:

•
$$I(Q) = I(0) \exp(-Q^2R_g^2/3)$$

•Ln[
$$I(Q)$$
].vs. Q^2 plot where $Q.R_q$ <1.0

•
$$R_{q} = \sqrt{(3.\text{slope})}$$

<u>Structure</u>

 $M \sim V \sim R^3$

•M=(1000.I(0).d².Na)/(C.
$$\Delta \rho^2$$
)

•Power-law at high q

Real Space

Smooth Surface



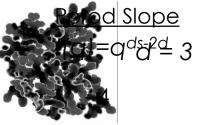
Rough Surface

Scaling Relation

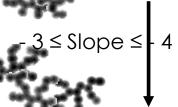


 $d_m = 3$

$$2 < d_S \le 3$$



Dispersion



Mass Fractal



$$1 \le d_S = d_m \le 3$$

Lilin He, et al. **Polymer**, Volume 105, 25-34 (November 2016) https://doi.org/10.1016/j.polymer.2016.10.019

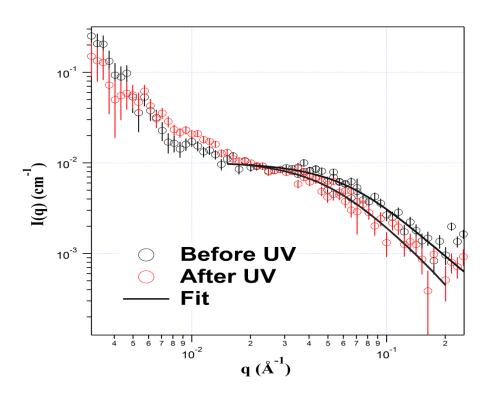




Fractal Dimension = d

Dale Shaefer: Nxschool talk 2017

SANS results: Unimer State



- 0.02wt%
- 15 ℃

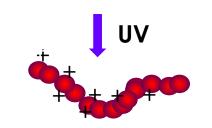
A model for polymers with excluded volume fraction yields Rg and Porod exponent

Before UV

 $R_g = 25.5$ Å Porod exponent = 1.94

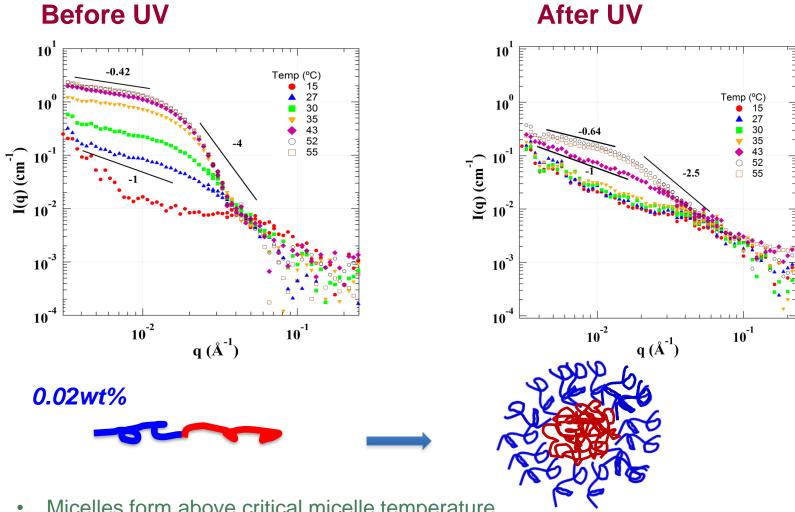
After UV

 $R_g = 31.1$ Å Porod exponent = 1.59



- Both solutions contain single chains and loosely-assembled clusters
- The chains are more stretched after UV irradiation

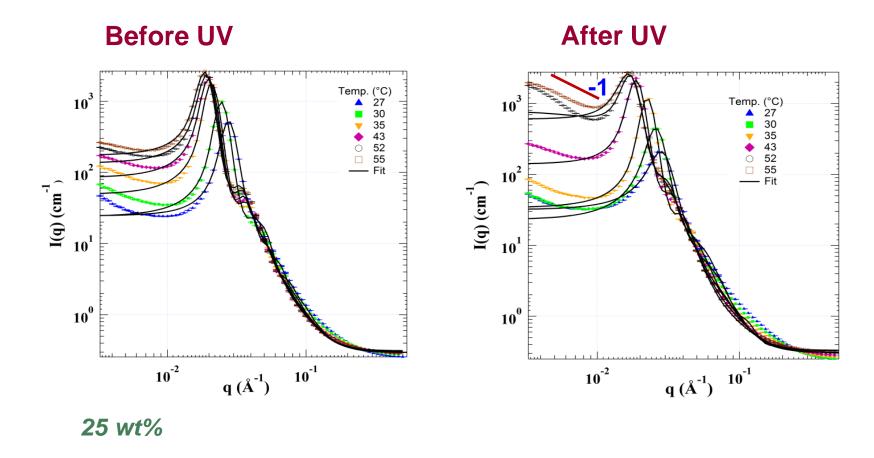
Temperature Effect: dilute solution



- Micelles form above critical micelle temperature
- Cleaving the light sensitive group defers the formation of the micelles

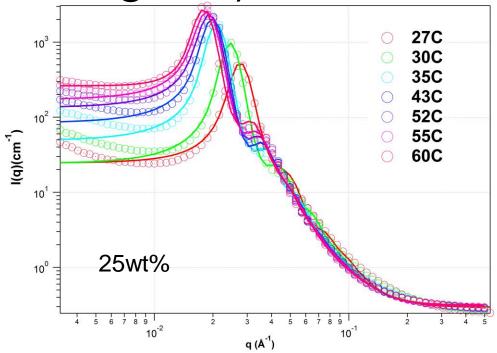


Temperature Effect: Concentrated solution



- Micelles grows with temperature;
- Possible sphere to rod transition at 55C for the UV exposed sample.

Model Fitting: Polycoreshell model



Material Balance Calculation

$$N_{ag}(131*v_{TN}+113*v_{PEO}*f)+30*c = \frac{4\pi}{3}R_A^3$$

$$N_{ag}$$
113 v_{PEO} *(1- t)+30*d = $\frac{4\pi}{3}(R_B{}^3 - R_A{}^3)$

$$N_{ag}(131*b_{TN}+113*b_{PEO}*f)+191*c = \frac{4\pi}{3}R_A^3*\rho_A$$

$$N_{ag}$$
113 b_{PEO} *(1- I)+191* $d = \frac{4\pi}{3}(R_B^3 - R_A^3) * \rho_B$

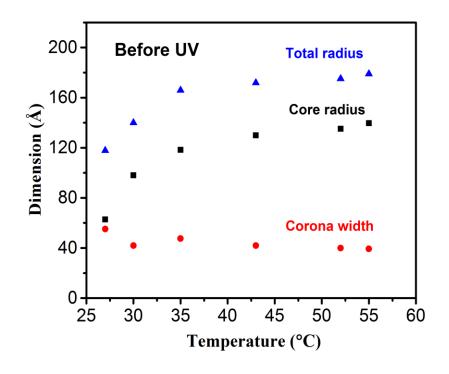


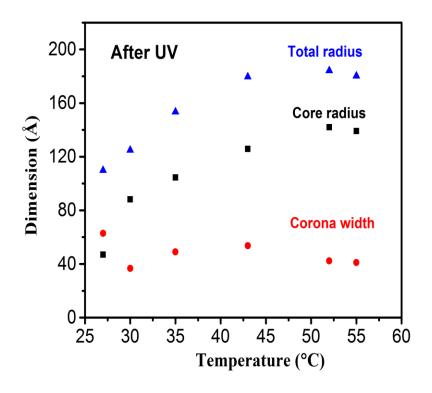
$$\frac{d\Sigma(Q)}{d\Omega} = \Delta \rho^2 \phi V_p P(Q) S(Q)$$

Percus-Yevick closure of the Ornstein-Zernike equation

- Micelle size (core and corona) and volume fraction
- Size polydispersity
- **SLDs** in core and corona
- **Aggregation number**
- Number of D₂O molecules in core and corona
- Fraction of PEO chains in core and corona

Temperature Effect: Concentration solution

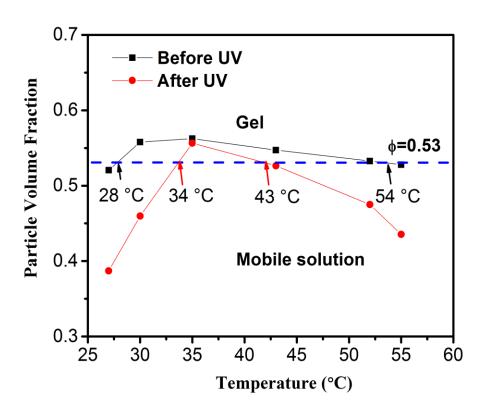




- The core size increases while the corona width remains;
- The slight shrinkage of corona at high temperatures is attributed to the loss of water,
 which is caused by the reduced miscibility of water and PEO.



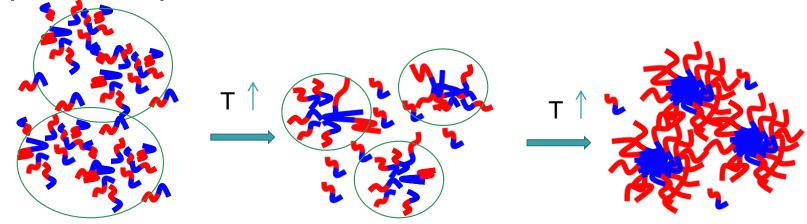
Temperature dependence of particle volume fraction



- Micelle effective volume fraction agrees the viscoelastic properties of solutions;
- Delayed gel formation in irradiated sample due to higher LCST transition;
- Critical value 0.53+/-0.02



Summary of Polymer Micellization

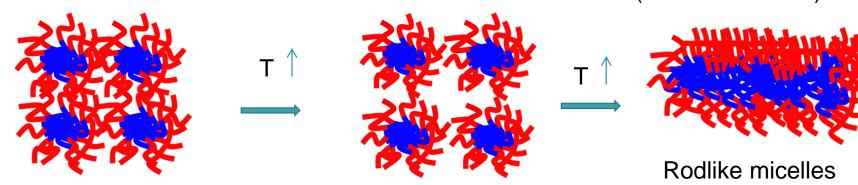


Unimers and clusters

Mixture of micelles and unimers

"well-defined" micelles (Micellar solution)

(for irradiated



Ordered micelles (Micellar gel)

Micellar solution sample only) (Less overlapping and entangled chains)



Realization of magnetic skyrmions in thin films at ambient conditions

Ryan Desautels, **Lisa DeBeer-Schmitt**, Sergio Montoya, Julie Borchers, Soong-Guen Je, Nan Tang, Mi-Young Im, Micheal Fitzsimmons, Eric Fullerton, Dustin Gilbert



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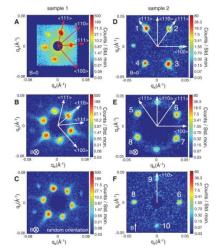




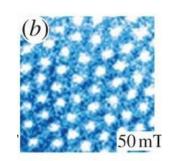
Skyrmions: a brief introduction



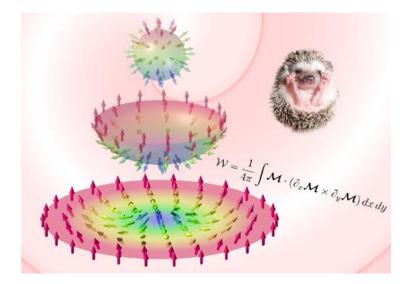
Tony Skyrme Nuc. Phys. 13, 556 (1962)



Science 323, 915 (2009)

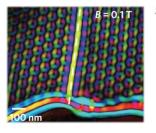


Nature 465, 901 (2010)



Predicted in magnetic systems

OAK RIDGE | HIGH FLUX | SOTOPE REACTOR | REAC

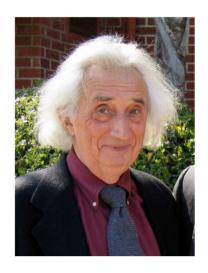


T = 260 K

Nature Mater. 10, 106 (2011)

Direct Exchange Prefers
Parallel Alignment

$$E \propto J(\overrightarrow{S_1} \cdot \overrightarrow{S_2})$$

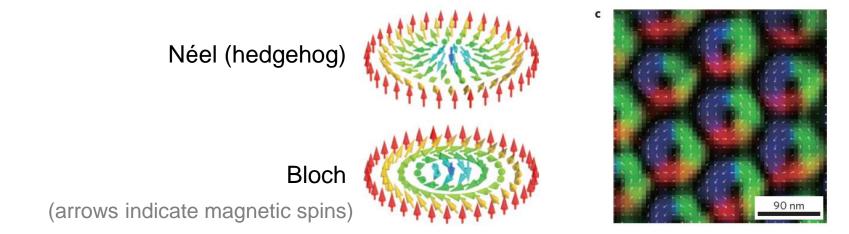


Igor Dzyaloshinskii Soviet Physocs Jetp. 5, 1259 (1957) Toru Moriya Phys. Rev. 120, 91 (1960)

$$E \propto D(S_1 \times S_2)$$

DM Interaction Prefers 90° Orientation Defines a 'handedness'

Skyrmions: a brief introduction



Circularity (CW and CCW)

Polarity (Core-up, Core-down) anti-parallel to perimeter

Genus 0 Genus 1 Genus 2 Genus 3 or more

Marble Doughnut Kettle Strainer

Bowling Ball Coffee Cup Scissors Grater

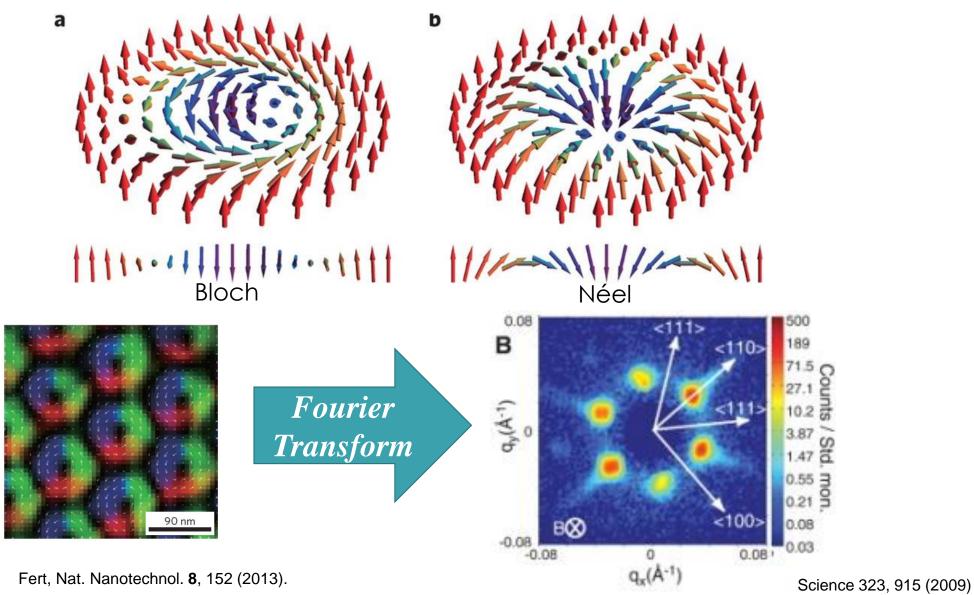
Fert, Cros, Sampaio, Nat. Nanotechnol. 8, 152 (2013).

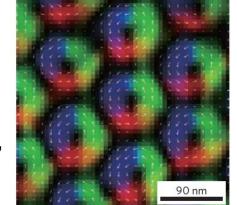
Changing between topological classes requires irreversible processes

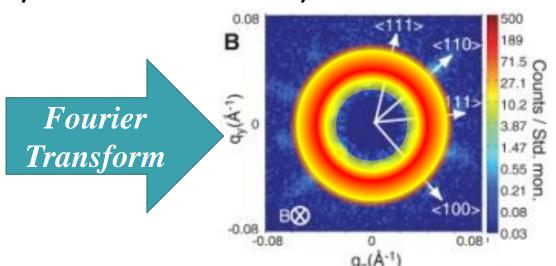
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Skyrmions: a brief introduction

I. Kezsmarki et al, Nature Materials **volume14**, pages1116–1122 (2015)





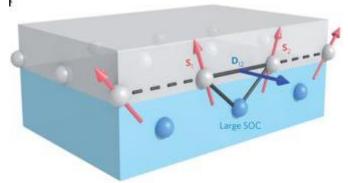


Fert, Nat. Nanotechnol. **8**, 152 (2013).

Imagine if we didn't have this long-range ordering...

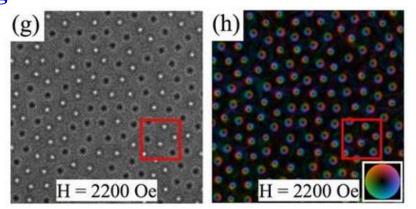
Origin of the long-range order is in magneto-crystalline coupling

We are interested in skyrmion systems which have weak magnetocrystalline coupling



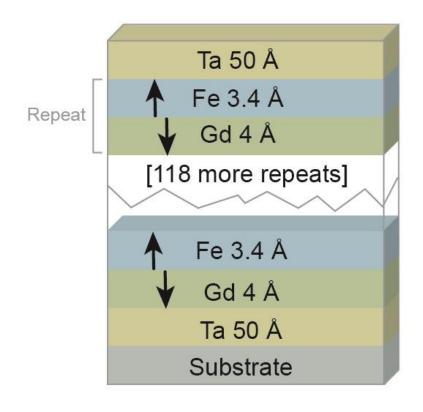
Fert et al., Nat. Nanotechnol. 8, 152 (2013)

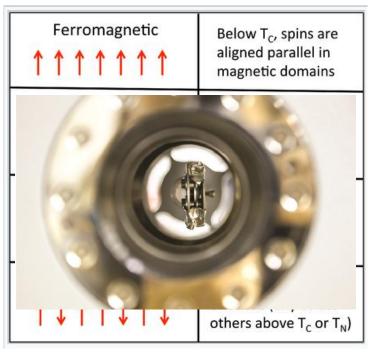




Montoya et al., Phys. Rev. B 95, 024415 (2017)

Fe/Gd Multilayer Thin Films: The Ingredients for Skyrmions at Ambient Conditions





chem.libretext.org

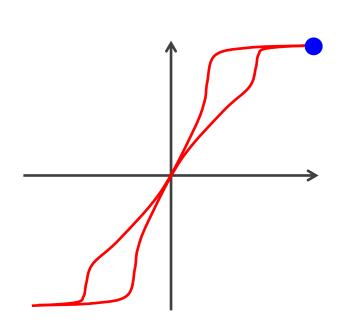
Ferrimagnetic construction with perpendicular magnetic anisotropy

Dipole-stabilized skyrmions

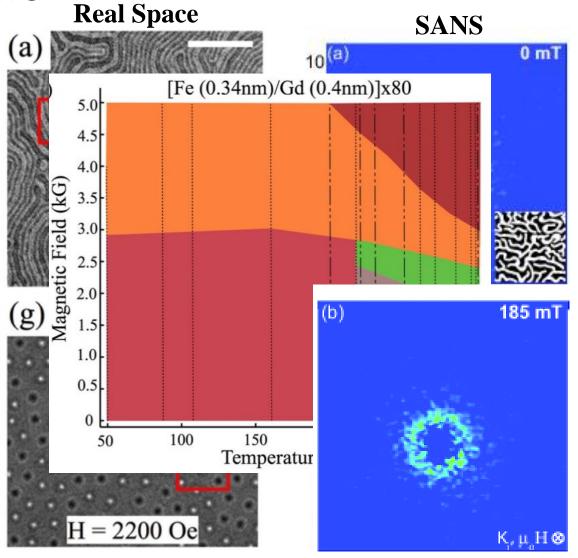
Limited/no DMI

No in-plane structure to cement a long-range orientation

Forming the Skyrmion State



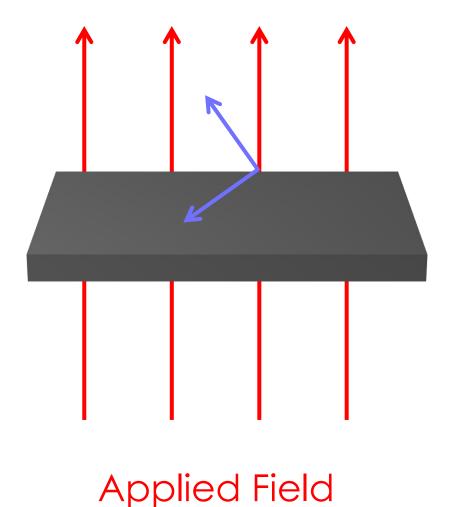
Montoya et al., Phys. Rev. B 95, 024415 (2017)

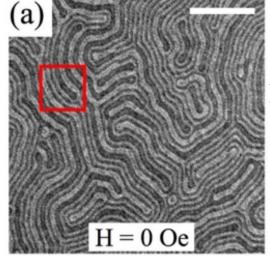


From saturation, worm domain remanent state

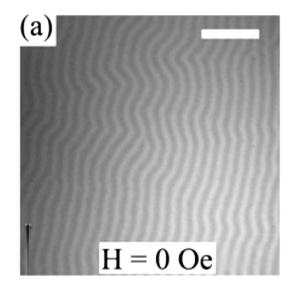
Break up into chiral bubbles, no chirality control, no long range order

Generating an Artificial Striped Phase





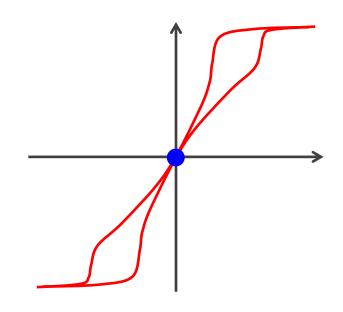
Labyrinth (or worm) domains

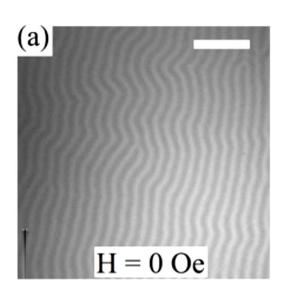


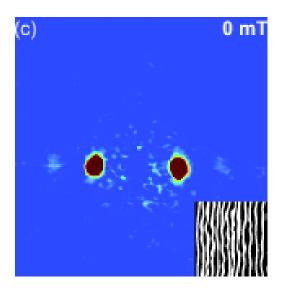
Tilting the sample imparts an in-plane field which breaks the symmetry

Artificial Stripe Domain

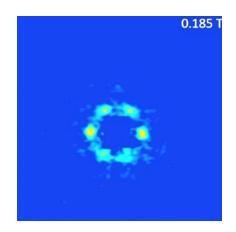
Generating an Artificial Striped Phase

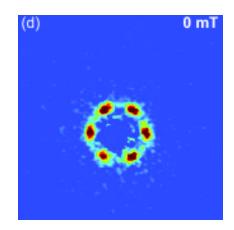


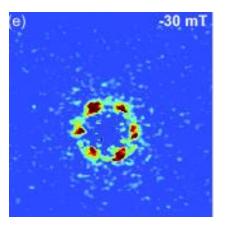


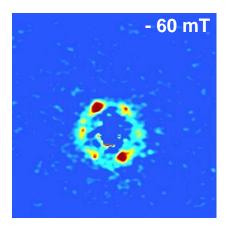


Desautels et al., Under Review

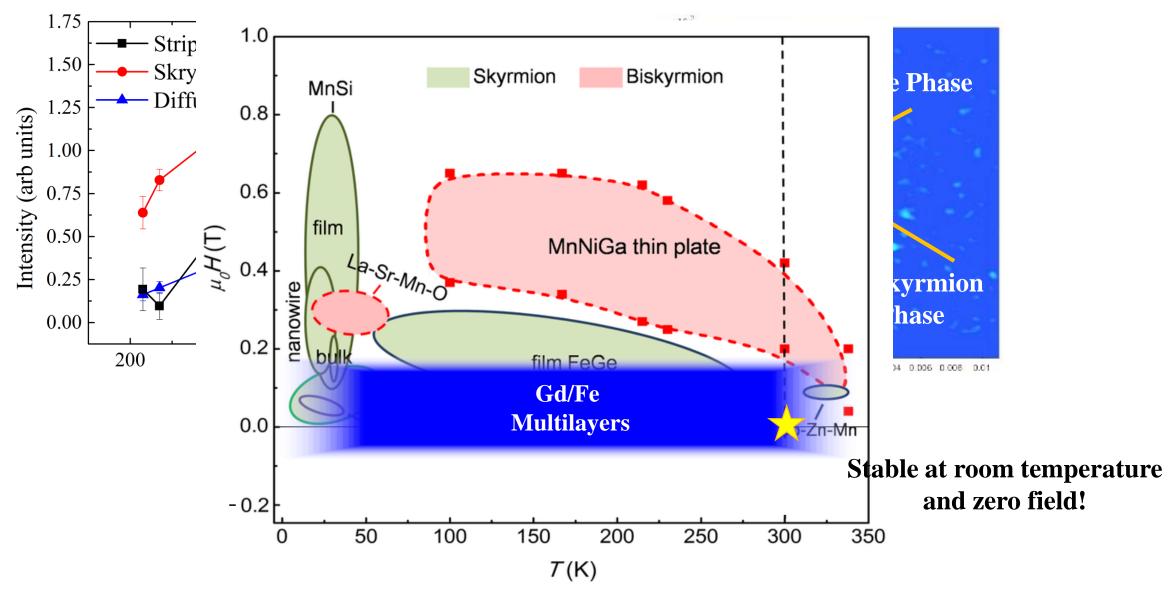






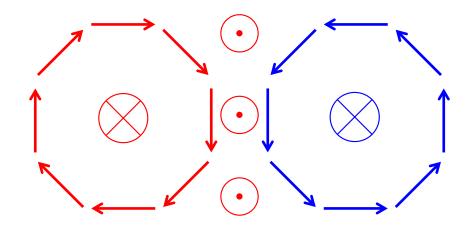


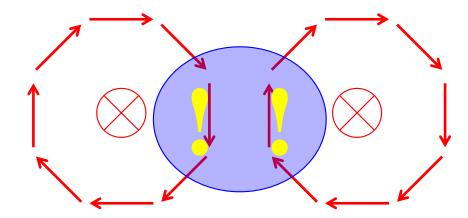
Skyrmion lattice Stability Envelope



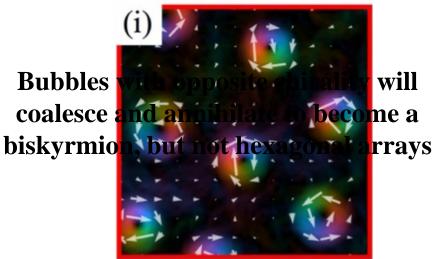


Implications of Chirality Control on the stability of Skyrmion Lattices





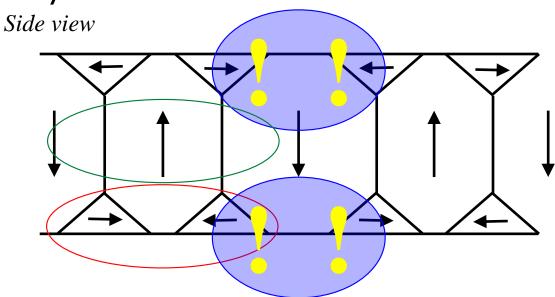
TEM indicates Bloch-type structure with no circularity control



Strong repulsive interactions between bubbles with identical chirality

→ Hexagonally Ordered Arrays

Implications of Chirality Control on the stability of Skyrmion Lattices



Appears as Néel Skyrmion, not Bloch (as observed with TEM)

Maybe Neel on the top and bottom, Bloch in the middle

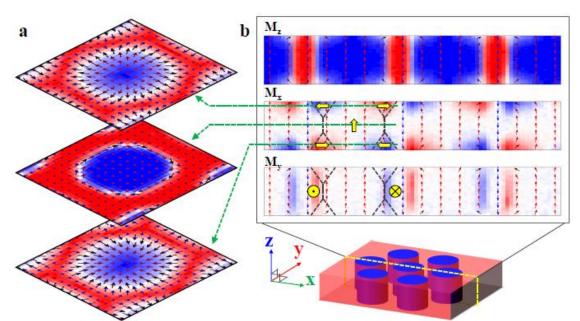
Without DMI, the Bloch region has no net chirality (as observed in TEM)

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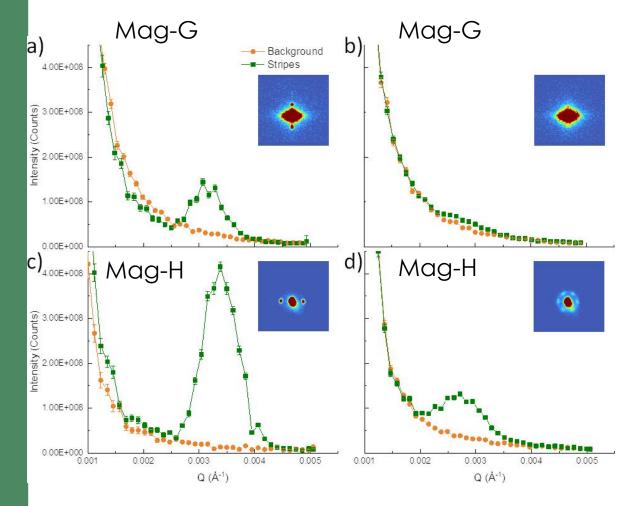
Assumed these are flux-closure domains with no defined chirality

Strong Repulsive Interaction

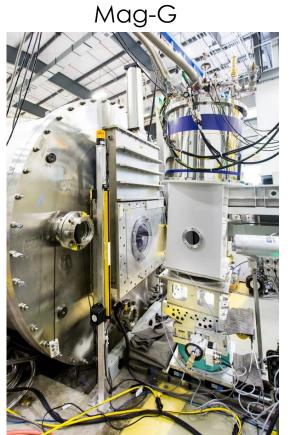
Repulsive interactions due to surface features, which have their chirality defined by the dipole fields



Thin Films studies at GP-SANS













Kevin G. Field¹, Kenneth C. Littrell^{1*}, and Samuel A. Briggs²

¹Oak Ridge National Laboratory

²Sandia National Laboratories

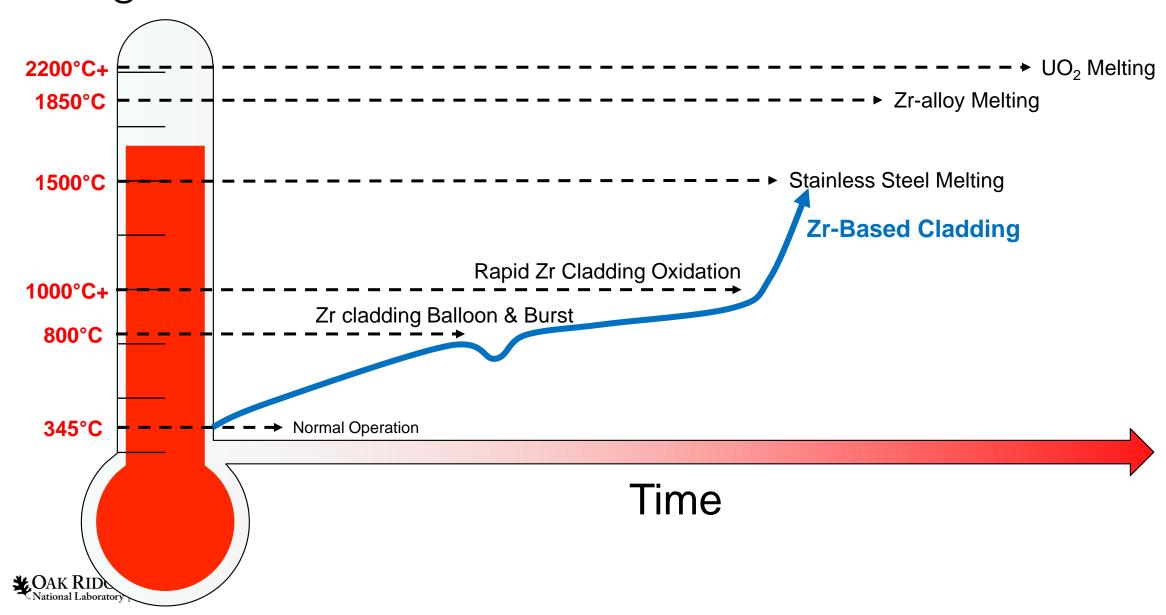


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Oxidation of cladding is key towards core degradation during a coolant-limited accident scenario



Oxidation of cladding is key towards core degradation during a coolant-limited accident scenario 2200°C+ Zr-alloy 1850°C Stainless Steel Me 1500°C APM, 1 bar steam Zirc-4, 1 bar steam 1500°C **Zr-Based Clac** 1500°C Rapid Zr Cladding Oxidation 1000°C+ Zr cladding Balloon & Burst 800°C **Oxidation Resistant Cladding** 345°C Normal Operation Time OAK RID

National Laborator

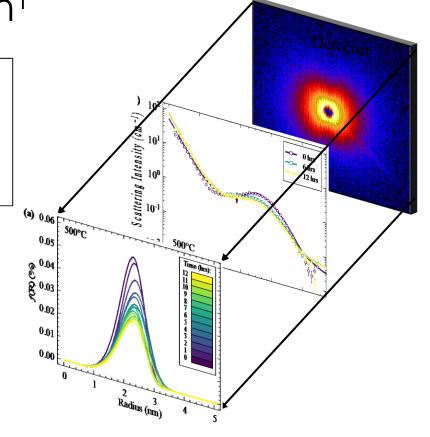
SANS Scattered Intensity: Local Monodispersed Approximation¹

$$\frac{d\sigma}{d\Omega}(q) = \Delta \rho^2 \int_0^\infty \phi(q,R)^2 S[q,R_{HS}(R)] N(R) dR + Aq^{-B} + C$$

- Contrast: $\Delta \rho^2 = (\rho_{particle} \rho_{matrix})^2$
- Form factor (for spheres):

$$\phi(q,R) = 3V_0[\sin(qR) - qR\cos(qR)]/(qR)^3$$

- Structure factor: $S[q, R_{HS}(R)] = [1 + 24\eta_{HS}G(R_{HS}q)/(R_{HS}q)^{-1}$
- Size distribution (Weibull density function): $N(R) = (R/\bar{R})^{b-1} exp[-(R/\bar{R})^b]$
- Low-q power-law: Aq^{-B}
- Background: C



J.S. Pedersen, Determination of Size Distributions from Small-Angle Scattering Data for Systems with Effective Hard-Sphere Interactions, *J. Appl. Cryst.*, **27**, pgs. 595-608, 1994.

Magnetic Shielded-SANS measurements:

Contrast from Neutrons

1. Nuclear contrast:

$$\Delta \rho_{nucl}^2 = \left(\rho_{nucl,particle} - \rho_{nucl,matrix}\right)^2$$

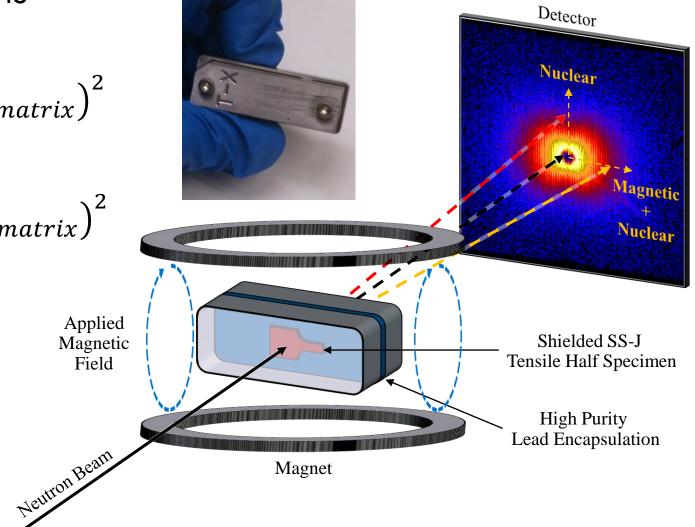
2. Magnetic contrast:

$$\Delta \rho_{mag}^2 = \left(\rho_{mag,particle} - \rho_{mag,matrix}\right)^2$$

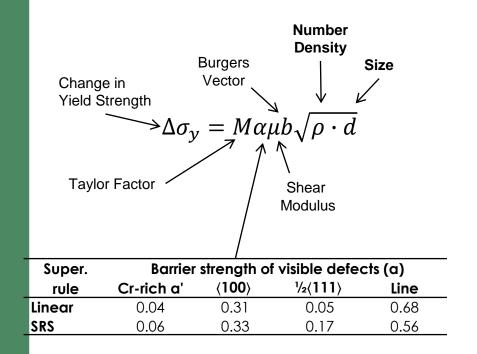
$$\frac{d\sigma}{d\Omega}(q) = \Delta \rho^2 \int_0^\infty \phi(q, R)^2 \dots []$$

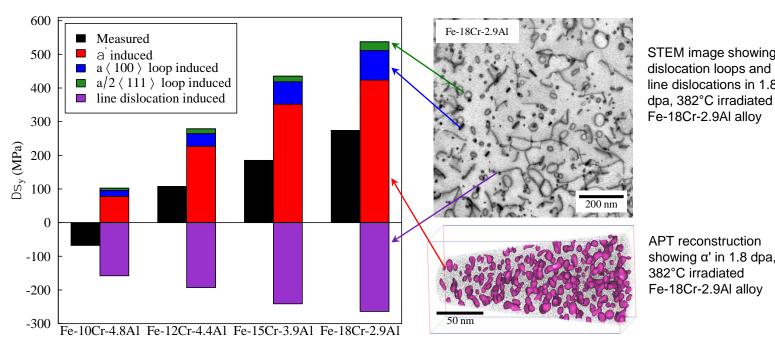
 $\left[\Delta \rho_{nucl}^2 + \Delta \rho_{mag}^2 sin^2 \alpha\right]$

Can exploit magnetization to extract composition



Radiation tolerance of FeCrAl alloys is analogous to FeCr alloys under similar irradiation conditions





dislocation loops and line dislocations in 1.8 dpa, 382°C irradiated Fe-18Cr-2.9Al alloy

APT reconstruction showing α' in 1.8 dpa, 382°C irradiated Fe-18Cr-2.9Al alloy

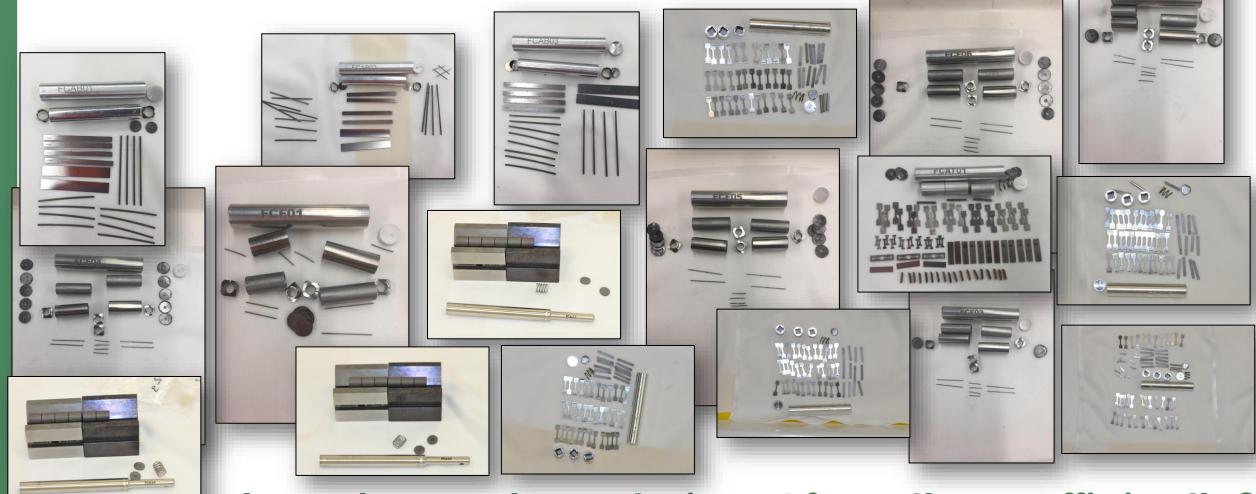
Dispersed barrier hardening (DBH) has linked radiation-hardening in FeCrAl alloys to formation of Cr-rich α' and dislocation loops after neutron irradiation to 1.8 dpa at 382°C

Need exists to understand the role of composition and temperature on the formation and progression of Cr-rich α' in FeCrAl alloys



A large amount of FeCrAl samples have been irradiated

and/or tested over the past 5 years...

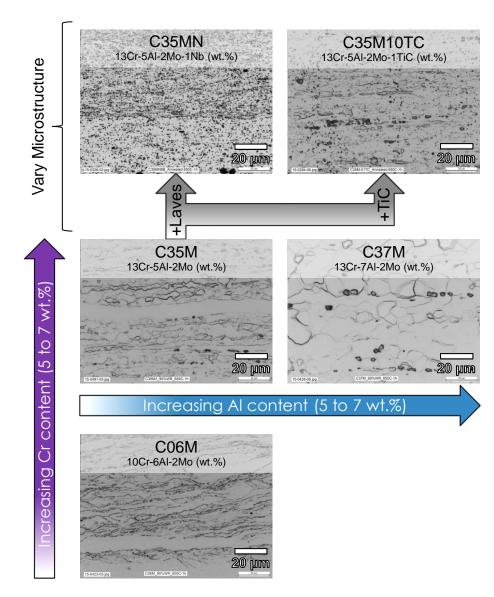


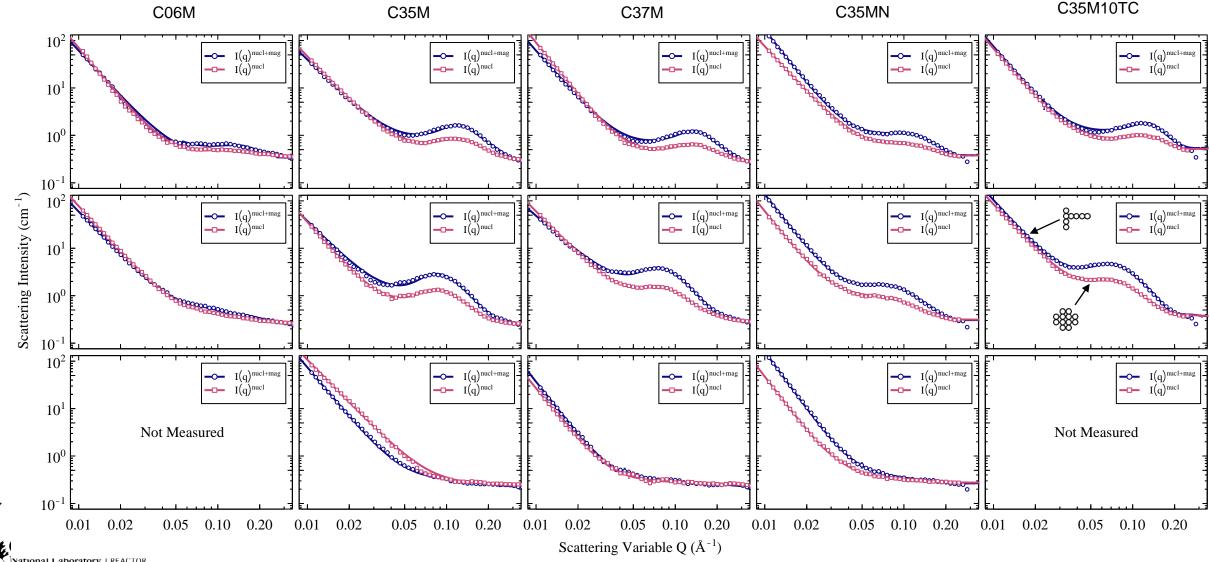
...how do we characterize a' from these efficiently?

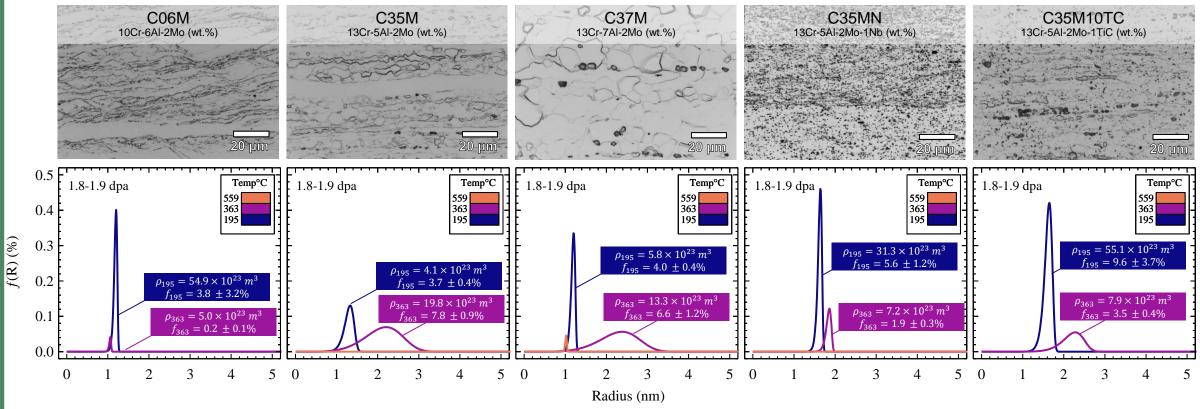
 Selected FeCrAl alloys irradiated in HFIR to determine temperature, composition, and microstructure trends

Capsule ID	Number Samples	Neutron Flux (n/cm²s) E > 0.1 MeV	Neutron Fluence (n/cm²) E > 0.1 MeV	Dose Rate (dpa/s)	Dose (dpa)	Irradiation Temperatur e (°C)
FCAT-01	45	1.10 × 10 ¹⁵	2.17× 10 ²¹	9.6× 10 ⁻⁷	1.9	194.5 ± 37.9
FCAT-02	45	1.04× 10 ¹⁵	2.05× 10 ²¹	9.1× 10 ⁻⁷	1.8	362.7 ± 21.2
FCAT-03	45	1.10× 10 ¹⁵	2.17× 10 ²¹	9.6× 10 ⁻⁷	1.8	559.4 ± 28.1

- Measurements performed at CG-2 general purpose SANS beamline at HFIR
- Data collected on broken tensile heads in magnetic shielded SANS configuration





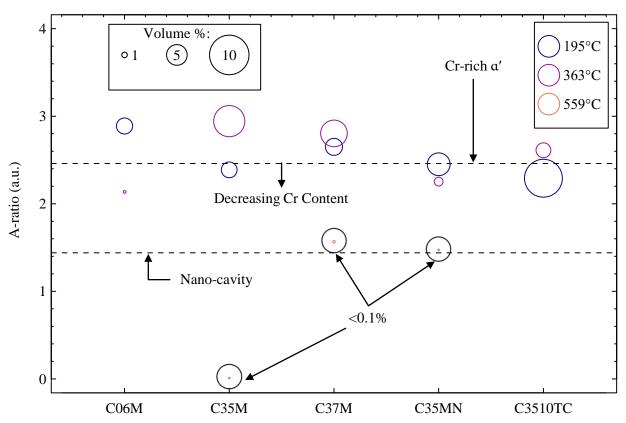


- Increasing irradiation temperature increases mean radius and distribution width
- Cr stronger than Al in control in α' precipitation
- Microstructure (precipitates/grain size) can play a role in α' precipitation

$$\frac{d\sigma}{d\Omega}(q) = \Delta \rho^2 \int_0^\infty \phi(q, R)^2 \dots []$$

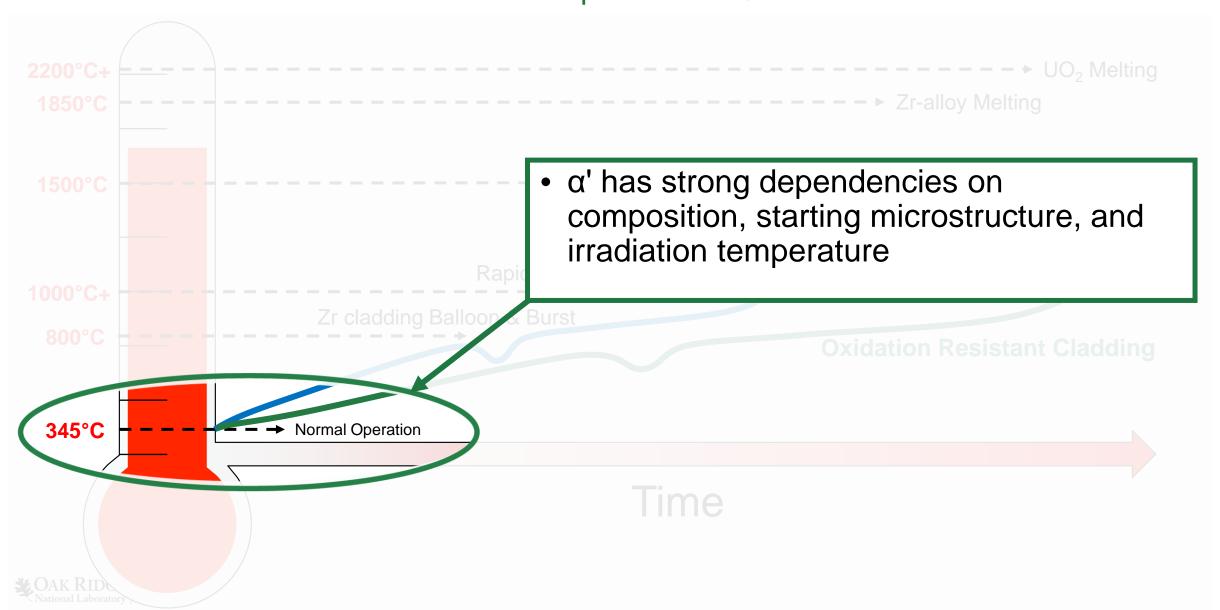
$$\left[\Delta \rho_{nucl}^2 + \Delta \rho_{mag}^2 \sin^2 \alpha\right]$$

$$A - ratio = \frac{\Delta \rho_{mag+nucl}^2}{\Delta \rho_{nucl}^2}$$



- Magnetic shielded SANS reveals that irradiations below 500°C form primarily α'
 - Typically lower Cr content in 195°C irradiations
- Above 500°C, scattering could be from nano-cavity formation

Practical Application of Ex-situ SANS results: a' concern for normal operation, but can be tuned



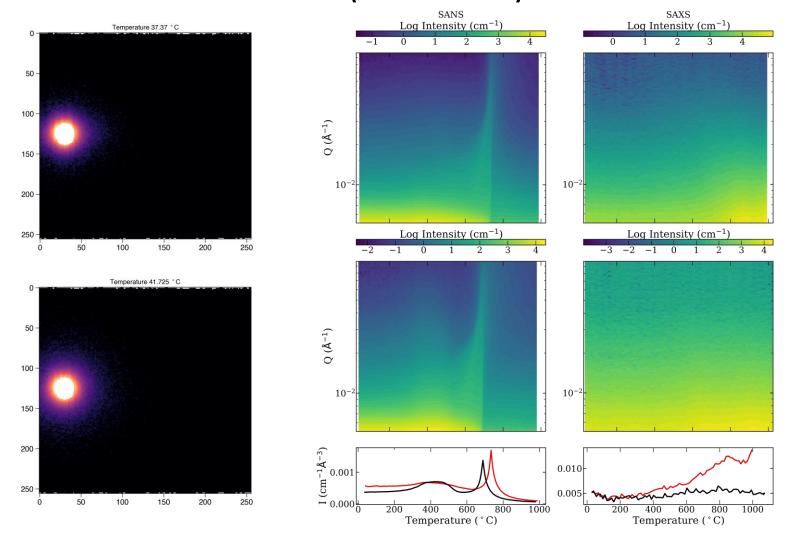


New Directions with SANS

- Time-resolved SANS
- GI-SANS



Time-resolved SANS (TR-SANS)



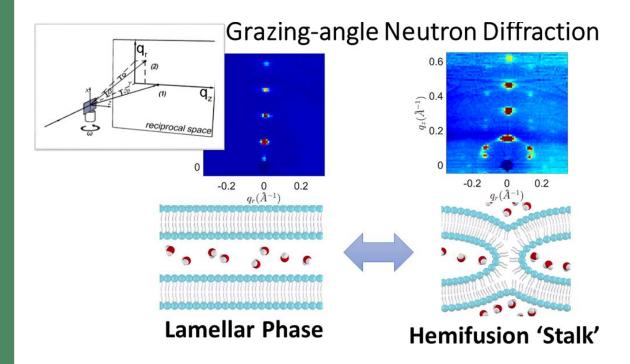
- Additively manufactured steels
- Data collected during ramp, 5 hour anneal, and cooling(not shown)
- SANS and SAX data taken under same heat treatment
- SANS sees metastable magnetic domain changes, but SAXS is only sensitive to the microstructure evolution

C. Fancher et al. In preparation



GI-SANS

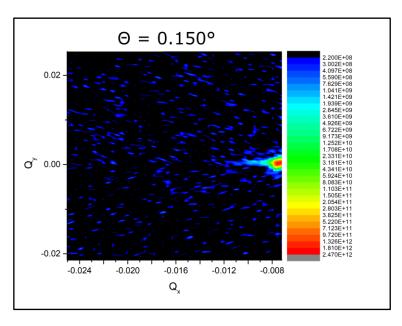
- Being developed at ORNL at EQ-SANS but GP-SANS, BIO-SANS can do this technique.
- Measuring the specular and off-specular versus sample angle encodes both in-plane and out-of-plane structure of the systems.



S. Qian and D. K. Rai J. Phys. Chem. Lett. 2018, 9, 5778-5784



Fe/Gd multilayers measured with a large (195 mT) out-of-plane field in the skyrmion phase



WLNC Liyanage et. al in preparation

Experts at ORNL on GI-SANS:

- Shuo Qian
- Changwoo Do
- Valeria Lauter

Thank you for your attention.

- SANS is powerful tool for probing mesoscale structures in a multitude of systems
- ORNL SANS have a wide variety of sample environments to enable your science

Questions?

Instrument Teams:

GP-SANS



Lisa DeBeer-Schmitt



Ken Littrell



Lilin He



Cody Pratt

Bio-SANS

EQ-SANS



Venky Pingali



Wellington Leite



Volker Urban



Luke Heroux

USANS



Wei-ren Chen



Yingrui Shang



Carrie Gao



Changwoo Do



William Heller



Gergely Nagy

National Laboratory REACTOR



